

# Selectively oriented crystalline growth of Mn<sub>3</sub>Sn on Al<sub>2</sub>O<sub>3</sub>(0001) using molecular beam epitaxy

S. Upadhyay,<sup>1</sup> T. Erickson,<sup>1</sup> H. Hannah,<sup>1</sup> A. Shrestha,<sup>1</sup> J. C. Moreno,<sup>2</sup> D. C. Ingram,<sup>1</sup> K. Sun,<sup>3</sup> A. R. Smith<sup>1+</sup>

<sup>1</sup> Nanoscale Quantum Phenomena Institute, Department of Physics and Astronomy, Ohio University Athens OH-45701, USA

<sup>2</sup> Universidad Autónoma de Puebla, Instituto de Física, Apartado Postal J-48, Puebla 72570, Mexico

<sup>3</sup> Michigan Center for Materials and Characterization (MCMC) & Department of Material Science and Engineering, The University of Michigan Ann Arbor, MI 48109, USA.

Kagome antiferromagnet Mn<sub>3</sub>Sn has garnered attention due to the presence of interesting properties such as anomalous hall effect below 420K [1], Nernst effects, and presence of exchange bias [2]. Until now, most thin film growths have been conducted by using sputter deposition or by cleaving. Recently, Higo *et al.* reported perpendicular full switching of the Mn<sub>3</sub>Sn device which was grown using molecular beam epitaxy on a MgO (110) substrate having a thin W buffer layer. The grown film had a (01 $\bar{1}$ 0) orientation and a perpendicular magnetic anisotropy which was attributed to a tensile strain of 0.25% in the [21 $\bar{1}$ 0] direction enabling 100% switching of the perpendicularly oriented magnetic octupole [3]. This result indicates that the orientation and possible strain of the film can be key for film transport properties, and therefore it is of great interest to explore how one can achieve different crystalline film orientations of Mn<sub>3</sub>Sn.

In this talk, we demonstrate the synthesis of crystalline Mn<sub>3</sub>Sn on Al<sub>2</sub>O<sub>3</sub> (0001) without a buffer layer using molecular beam epitaxy. The samples were deposited at two different temperatures T<sub>h</sub> (500 ± 9 °C) and T<sub>l</sub> (416 ± 9 °C) with Mn: Sn flux ratio of 3.2:1 for 90 minutes. Our analysis indicates that for the two temperatures, the resulting orientations of the films are different, with the T<sub>h</sub> sample being predominantly *c*-oriented and T<sub>l</sub> sample being 43% *a*-oriented, but in our recent template growth method at room temperature, we achieved an 82% *a*-oriented film. In both cases, the reflection high energy electron diffraction patterns were streaky indicating a crystalline film, and the cross-sectional scanning tunneling electron microscopy gave an insight into the morphology of the samples. In both cases, the sample are discontinuous with a 3D morphology for the T<sub>h</sub> - grown sample and quasi-2D morphology for the T<sub>l</sub> - grown sample. Our recent template-grown samples are contiguous and show streaky RHEED patterns throughout the growth. Orientation relationships between the Mn<sub>3</sub>Sn films and the sapphire substrates are determined from *in-plane* and *out-of-plane* measurements. The composition of the samples, the strain effects, and the crystalline structures will be discussed in detail. Furthermore, we are in the process of doing STM measurements on the samples and performing theoretical calculations.

[1] Z. Zhao, Q. Guo, F. Chen, K. Zhang, and Y. Jiang. *Physica B: Condensed Matter*, **604**, 412692 (2021).

[2] X.F. Zhou, X.Z. Chen, Y.F. You, L.Y. Liao, H. Bai, R.Q. Zhang, Y.J. Zhou, H.Q. Wu, C. Song, and F. Pan, *Phys. Rev. Applied* **14**, 054037 (2020).

[3] T. Higo, K. Kondou, T. Nomoto, M. Shiga, S. Sakamoto, X. Chen, D.N.-Hamane, R. Arita, Y. Otani, S. Miwa and S. Nakatsuji, *Nature* **607**, 474-479 (2022).

+ Author for correspondence: [smitha2@ohio.edu](mailto:smitha2@ohio.edu)

## Supplementary Pages

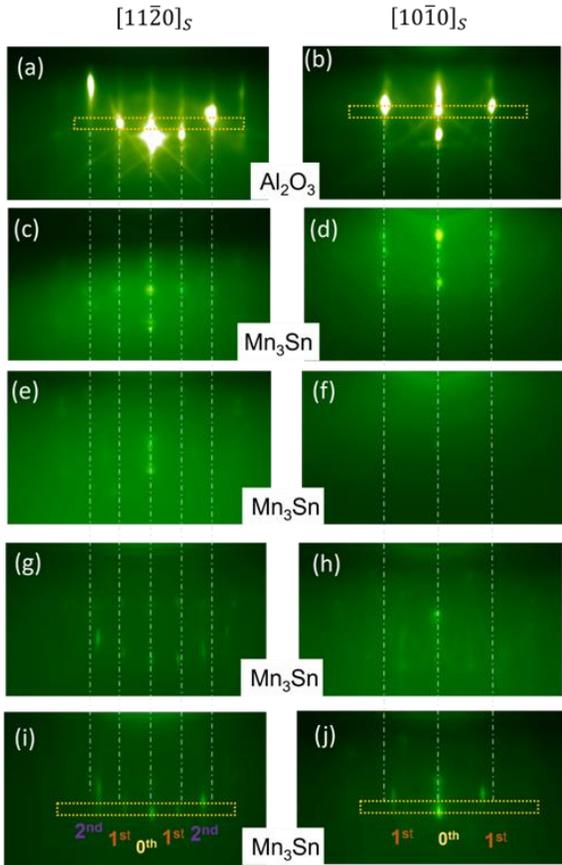


Figure 1: RHEED patterns of  $\text{Al}_2\text{O}_3$  and growth of  $\text{Mn}_3\text{Sn}$  for sample A grown at  $T_h$ ; (a-b) annealed  $\text{Al}_2\text{O}_3$ ; (c-d) immediately after opening the shutter; (e-f) after 90 mins of  $\text{Mn}_3\text{Sn}$  growth; (g-h)  $\text{Mn}_3\text{Sn}$  next day at 300 K; (i-j)  $\text{Mn}_3\text{Sn}$  after 5 days at 300 K.

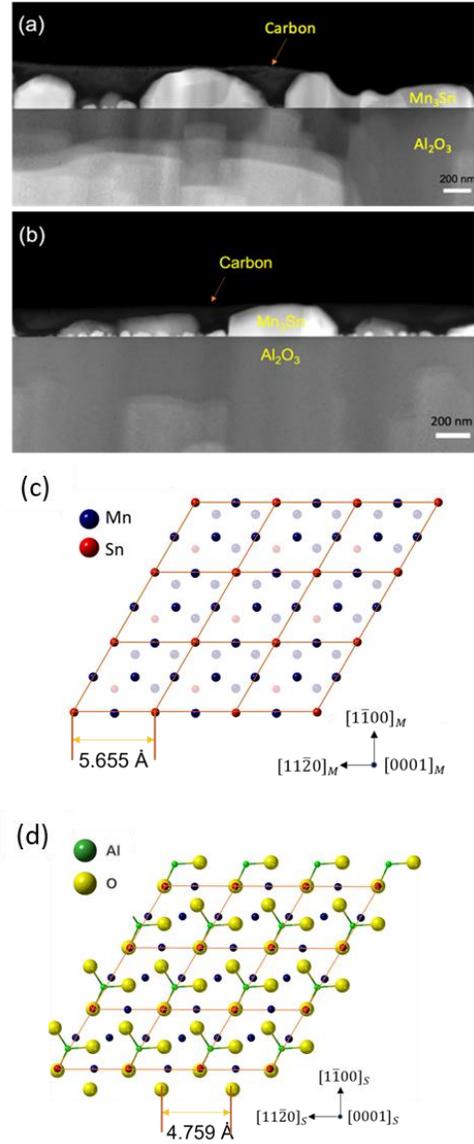


Figure 2: (a)-(b) STEM images taken at a high-angle annular dark field (HAADF) for sample A (grown at  $T_h$ ); (c) Top view of the 2D crystal for sample A. The faded atoms indicate the  $z = \frac{1}{2}$  position. The orange lines indicate the unit cells of  $\text{Mn}_3\text{Sn}$ ; (d) The overlay of  $\text{Mn}_3\text{Sn}$  on  $c$ -(0001)  $\text{Al}_2\text{O}_3$  is shown for sample A.

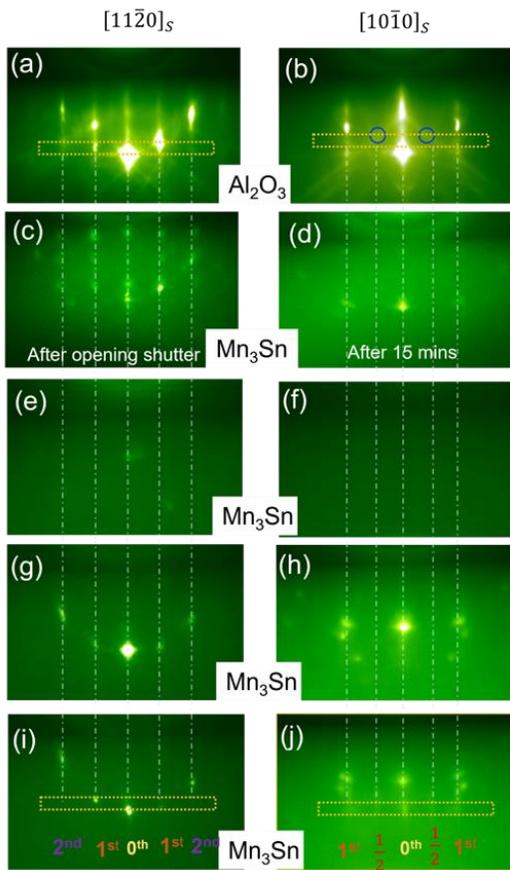


Figure 3: RHEED patterns of  $\text{Al}_2\text{O}_3$  and growth of  $\text{Mn}_3\text{Sn}$  for sample B grown at  $T_1$ ; (a-b) annealed  $\text{Al}_2\text{O}_3$ ; (c) immediately after opening Mn and Sn shutter simultaneously; (d) after 15 mins of growth; (e-f) after 90 mins of growth; (g-h)  $\text{Mn}_3\text{Sn}$  next day at 300 K; (i-j)  $\text{Mn}_3\text{Sn}$  after 2 days at 300 K.

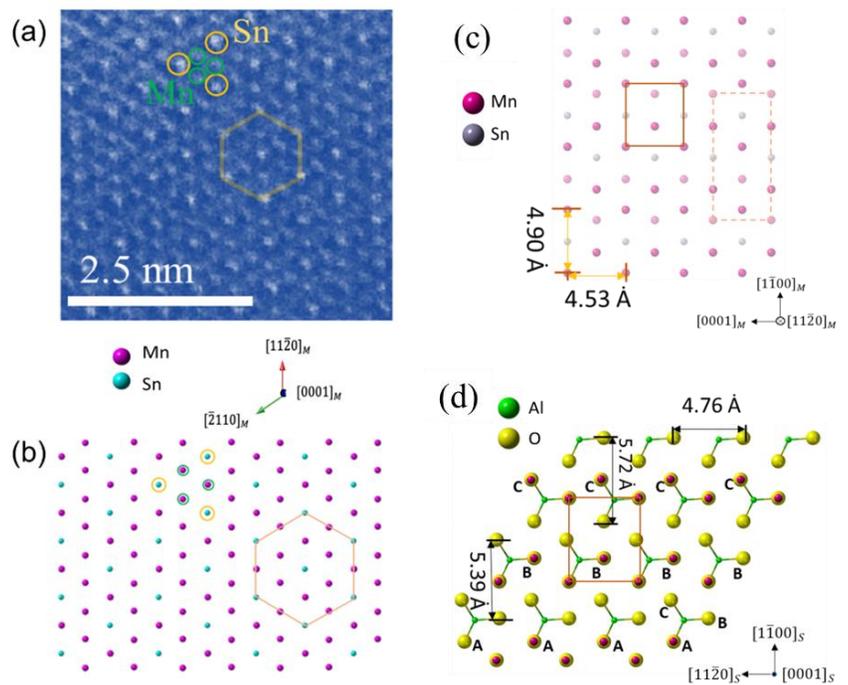


Figure 4: (a) Lattice imaging for sample B; (b) model showing the corresponding  $\text{Mn}_3\text{Sn}$   $c$ -plane lattice; (c) top view of the 2-D crystal model of the  $a$ -plane oriented sample and (d) the pseudomorphic overlay of  $\text{Mn}_3\text{Sn}$  on  $c$ -(0001)  $\text{Al}_2\text{O}_3$ .