ALD Fundamentals

Room Arteveldeforum & Pedro de Gante - Session AF-MoP

ALD Fundamentals Poster Session

AF-MOP-1 Unearthing Ruthenium Diazadienyls as Precursors for the ALD of Ru Thin Films: Precursor Assessment and ALD Process Development, *David Zanders, J. Obenlüneschloss,* Ruhr University Bochum, Germany; *M. Gock,* Heraeus, Germany; *A. Devi,* Ruhr University Bochum, Germany

Ruthenium thin films are garnering significant interest as next-generation interconnects to replace Cu in future nanoelectronic devices. Especially in the back end of line (BEOL) and middle of line (MOL), ongoing scale down towards 2 nm and beyond has motivated alternative metallization approaches such as semi-damascene in which Ru outperforms Cu and Co notably.^[1,2]One deposition technique most suitable to provide thin films for challenging interconnect device architectures is ALD.^[3] A paramount factor in each ALD process is the choice of precursors and their chemistry that governs layer formation and material quality. A review of the current ALD processes for Ru thin films shows that a considerable number of often closely related precursors with their individual advantages and drawbacks have been employed.^[4] Hitherto, none of them could fully satisfy academic and industrial demands alike.

This presentation describes the synthesis and detailed characterization of an alternative Ru precursor class: Ru diazadienyl cymenes [Ru(DAD)(Cym)] and their potential for CVD and ALD applications. Two examples, Ru(tBuDAD)(Cym) and Ru(iPrDAD)(Cym) (Figure 1) were obtained in high yields in a one-pot synthesis and thoroughly assessed in terms of purity (NMR, EA) and structure (SC-XRD). The structures were reproduced by DFT studies and subjected to bond dissociation analysis which rendered them appreciably stable. Thermal stability upon evaporation was evaluated in more detail alongside overall evaporation behaviour in thermogravimetric analyses (TGA) (Figure 2). Vapor pressures of the two complexes were derived and compared to the most often used Ru(EtCp)₂. Especially Ru(iPrDAD)(Cym) showed appreciable volatilization and deemed competitive towards the reference compounds.

Thus, Ru(iPrDAD)(Cym) was employed as precursor in a PEALD process with O₂ plasma. Preliminary results demonstrated the growth of thin, pinhole-free, low roughness films on Si(100) at temperatures as low as 120 °C (Figure 3). A full process study investigating typical ALD growth characteristics in terms of saturation, temperature dependency of film growth and thickness scalability was carried out to assess the ALD behaviour. Complementarily, AFM, RBS/NRA and XPS as well as resistivity measurements were performed.

[1]G. Murdoch, et al., in 2020 IEEE IITC 1052020, p. 4.

[2]https://www.imec-int.com/en/press/imec-presents-alternative-metalsadvanced-interconnect-and-contact-schemes-path-2nm.

[3]M.-J. Li, et al., in 2021 IEEE IITC 762021, p. 1.

[4]https://www.atomiclimits.com/alddatabase/.

AF-MOP-2 Towards Greener ALD Processes: The Precursor Chemistry Is Key, *Matthieu Weber*, LMGP - Grenoble INP, France; *O. Graniel, D. Muñoz-Rojas*, LMGP - Grenoble INP, France; *N. Boysen, A. Devi*, Ruhr Universität Bochum, Germany

ALD is a versatile and emerging technology, allowing for the precise coating of challenging substrates with a nanometer control over thickness. Due to its unique assets, a substantial growth of the ALD market is expected. In the current context of moving towards greener processes, the ALD technique offers the potential to become greener and to tackle environmental challenges. Indeed, the process itself has currently a consequent impact on the environment which should ideally be reduced as the technique is implemented in a wider range of products and applications. Based on a literature overview, our findings show that the duration of the process, the temperature used, and the precursor chemistry are key factors affecting the environmental impact of ALD. The principles of green chemistry are discussed considering the specificities of the ALD process, and different ways to reduce the impact are proposed, in particular the optimization of the processing parameters, the use of spatial ALD (SALD) and the chemical design of greener precursors are shown as efficient routes to lower the ALD environmental impact and improve its sustainability.

AF-MoP-3 An IR Spectroscopy Study of the Degradation of Surface Bound Azido-Groups in High Vacuum, Sofie S. T. Vandenbroucke, M. Nisula, R. Petit, Ghent University, Belgium; R. Vos, K. Jans, P. Vereecken, IMEC, Belgium; J. Dendooven, C. Detavernier, Ghent University, Belgium

'Click chemistry' encompasses a set of powerful chemical reactions that have a high vield, are highly selective and specific, proceed under simple reaction conditions, use readily available starting materials, create stable products and harmless by-products.[Kolb, H.C., et al. (2001) Ang. Chem. Int.] The azide-alkyne cycloaddition reaction is considered the cream of the crop of the click reactions. Hence, surface functionalization by azidomoieties is often desired for applications in drug discovery, polymer chemistry, materials engineering and biosensor devices. [Lahann J. (2009) Click chemistry for biotechnology and materials science] This can be achieved by depositing self-assembled monolayers using azido-containing organosilanes which typically involves liquid-phase protocols. The use of vapour-phase chemistry might be beneficial regarding integration into a high-throughput production sequence and the deposition of high-quality and reproducible coatings on 3D microstructures. [Vos, R., et al. (2018) Langmuir] Because vapour-chemistry typically involves the use of high vacuum equipment, the stability of the azido-groups in high vacuum is of utmost importance.

In this work, [Vandenbroucke S.S.T. (2021) *Langmuir*] the stability of azidocontaining self-assembled monolayers is monitored in real-time using in situ reflection FTIR at a temperature of 150 °C and a pressure of 1E-5 mbar for 8 h. The data in Figure 1 displays a clear decrease of the asymmetric azide stretching vibration at 2105 cm⁻¹ over time in high vacuum, suggesting the degradation of the azido-groups. The degradation is further investigated at three different temperatures and seven different nitrogen partial pressures using ex situ ATR-FTIR. The degradation is found to increase at higher temperatures and lower nitrogen pressures. This is in accordance with the theory that the degradation reaction involves the decomposition into molecular nitrogen and the formation of a highly reactive nitrene.

Many applications in the semiconductor industry require a perfect control over the surface chemistry down to the nanometre level to yield reproducible results. For the condition with the most degradation only 63% of azides are found to remain at the surface after 8 h in high vacuum. This would imply a significant loss in control over the exact surface chemistry. One should therefore always consider the stability of functional groups such as azides when depositing or post-processing functional coatings in high vacuum.

AF-MoP-4 Nucleation of AlOx on ZrOx and ZrOx on AlOx studied by Low Energy Ion Scattering, Jan Tröger, Tascon GmbH, Germany; M. Näsi, T. Blomberg, Picosun Oy, Finland; R. ter Veen, Tascon GmbH, Germany

Nucleation is a crucial step in the formation of ALD films. It determines the formation of the first atomic layer of film. Proper nucleation is needed to deposit closed films with a homogeneous thickness. Since the nucleation deals with the formation of the first atomic layer of film, a very surface specific tool is needed to study nucleation: It should be able to distinguish between a closed monolayer and a half-closed double layer.

For common other techniques, such as XRF or XRR, it is not possibly to make this distinction due to a lack of surface specificity. Low Energy Ion Scattering (LEIS), on the other hand, is specific to the very first atomic layer, enabling it to determine the closure of the film (surface fractions of film and substrate), rather than the overall number of atoms per unit area that XRF or XRR provide.

We will demonstrate how LEIS works for the nucleation of AlOx films on ZrOx and the ZrOx films on AlOx. We show the normalized LEIS peak areas for Al and Zr as a function of cycle number for both depositions. The results show that in both cases, the films are almost closed after 20 cycles. In addition, we will present and discuss the following two features: In the deposition of ZrOx on AlOx, the peak area for Zr "overshoots". This indicates that the initial deposition happens with a stoichiometry that Zr-rich compared to ZrO_2 (and has some similarity to that of the Al_2O_3 substrate). In the case of the deposition of AlOx on ZrOx, the peak area for Al "undershoots", indicating that the initial deposition proceeds with an Alpoor stoichiometry compared to that of Al_2O_3 (with some similarity to that of the ZrO₂ substrate).

AF-MOP-5 Characteristics Evaluation of SiO₂ Thin Film Deposited by Organic vs Inorganic Precursor in ALD, Jihyeok An, I. Yang, S. Yoon, DUKSAN Techopia Co., Ltd, Korea (Republic of); I. Park, Hanyang University, Korea

As semiconductor device highly integrated in a chip, it required a more complex structure at unit devices such as transistor, capacitor, and resistor. The thin film coated in structure needs better thickness controllability and uniformity as well as very smooth surface, especially in the complicated 3D-structure. To fulfill the demand, ALD method, which provided excellent step coverage, has been strongly adopted in the fabrications of metal oxides and metal itself. Silicon oxide(SiO₂) has been the most representative material used for the longest time in semiconductor industry in gate, floating gate, and other processes. The film characteristics of SiO₂ strongly depend on the precursor type and process parameters even though it was made with the same ALD method.

In this work, the process and materials characteristics dependency of precursor type, especially organic or inorganic, have been studied in ALD process and film properties. SiO₂ thin films were deposited by plasma enhanced ALD system with respective organic (tetrakis(dimethylamino)silane, tDMAS) and inorganic (hexachlorodisilane, HCDS) precursor. ALD window region was confirmed for both precursors. The growth rate of SiO₂ films fabricated with tDMAS and HCDS were 0.78 and 0.21 Å/cycle at the deposition temperature of 400°C and 300°C, respectively. The Si : O ratio of SiO₂ film was 1.9 for tDMAS and 2.0 for HCDS. The density of both films was the same as 2.5. The surface of SiO2 film made with tDMAS, 0.1 nm RMS roughness, is smoother than that with HCDS, 0.5 nm. The MOS devices based on both SiO₂ films were manufactured with Si substrate and TiN metal electrode, and their capacitance properties were analyzed. The dielectric constant of both films showed the same value of 3.9.

AF-MoP-6 Microstructure and Electrical Properties of Conductive Nitride Films Grown by Plasma Enhanced Atomic Layer Deposition with Considerable Ion Bombardment, I. Krylov, Tower Semiconductor Ltd., Israel; Valentina Korchnoy, X. Xu, K. Weinfeld, E. Yalon, D. Ritter, M. Eizenberg, Technion - Israel Institute of Technology, Israel

We investigated electrical properties and structure of conductive nitrides (TiNx, ZrNx, TaNx, HfNx) deposited by the PEALD process from amide precursors and H₂/Ar plasma. Ion bombardment introduces additional energy to the growing film promoting chemical reactions and film densification. In PEALD process involving H₂/Ar plasma, most of bombardment effects are related to heavier Ar ions which are responsible for obtaining highly crystalline nitride films. Ion energy depends on the pressure inside ALD reactor during plasma half-cycle. Low pressure is favorable for deposition of highly conductive nitrides. The deposited nitrides have different grain size, carbon contamination, and variation of the film density from theoretical value despite same process conditions. Films crystallinity was investigated by the HAADF-STEM (Fig.1). The deposited nitride films differ in film morphology. TiNx films consist of large columnar grains elongated through the whole film thickness. HfNx films has columnar-like grains with smaller grain size compared to TiN_x . ZrN_x and TaN_x films consist of small grains. Significant ion bombardment reached in the process results in the growth of (111)-oriented polycrystalline films with low O contamination (Table1). The films are polycrystalline (Fig.2) despite the significant carbon content. Crystallization is stimulated by the ion bombardment effect achieved by the lowing plasma pressure. Carbon contamination correlates with the level of C-content in the precursor molecule.

Electrical resistivity of deposited nitrides is summarized in Fig.3,a. Despite near similar bulk resistivity of TiN, ZrN, and HfN, a significant difference in resistivity was found in the deposited films. Among the investigated films, TiN_x has the highest grain size and lowest resistivity. Density of the deposited nitrides was measured by the XRR. Deviation of the film density from its theoretical values represents the degree of crystallinity at a given film and follows the resistivity trend. Higher deviation indicates the smaller grain size resulting in higher electron scattering at grain boundaries. Film density of transition metal nitride deposited by PEALD. Positive TCR values (metallic) were obtained for TiN_x and ZrN_x films, whereas the deposited TaN_x and HfN_x films have negative TCR values (semiconducting) (Fig.3,b). Negative TCR values are attributed to the presence of nitrogen-rich phases, MN_x > 1 in these films. Among the investigated films, TiN_x films have the

highest grain size and lowest resistivity. Therefore, TiN_x is proposed as the leading PEALD nitride for conductive applications.

AF-MoP-7 Room-Temperature Atomic Layer Deposition of Iron Oxide and Its Surface Reaction; Explained by *in-Situ* Observation, *K. Yoshida, I. Nagata, K. Saito, M. Miura, K. Kanomata, Fumihiko Hirose,* Yamagata University, Japan

Room-temperature (23- 25°C) atomic layer deposition (RT-ALD) of iron oxide is developed with a precursor of bis(N, N'-diisopropylpropionamidinate)iron ((DIPPA)₂Fe) and plasma excited humidified Ar. Iron oxide such as Fe₂O₃ is examined as a magnetic coating material for pharmaceutical powders in drug delivery. Since the ALD uses saturation of the surface reaction, it is possible to deposit the oxide film on powder surfaces with nanometer precision. We consider that the ALD could be applied to heat-sensitive pharmaceutical drugs, although there were no previous reports on the RT Fe₂O₃ ALD. To find the gas injection conditions in ALD, we investigate the saturation of precursor adsorption and oxidation on the iron oxide surface in this study. The ALD growth is confirmed by quartz crystal microbalance (QCM). We used plasma excited humidified Ar as an oxidizing gas, and (DIPPA)₂Fe as precursors of iron oxide. Any heaters or thermal sources were not installed on the ALD chamber wall. The (DIPPA)₂Fe container and gas delivery pipe were warmed up around 80 °C to promote the precursor evaporation without its condensation. The exposure of Fe precursor in the adsorption step was 2.0× 10⁵ Langmuir (L). The oxidation time was set at 600 s. The evacuation steps of 300 and 60 s were inserted between the precursor and the plasma excited humidified Ar injections. P-type Si (100) substrates and polyethylene naphthalate (PEN) films were used for the ALD tests. We confirmed the film composition by XPS with a MgK α X-ray source. For the QCM measurement, we used an ALD sensor (#750-717-G2, Inficon) with AT-cut quartz crystals in which the resonance frequency was 6 MHz. The sampling rate was 2 Hz. We examined the iron oxide deposition on Si(100) and PEN. A sample image of the iron oxide coated PEN is shown in figure 1. The sample is colored brown by the interference of iron oxide thin film, but peelings and cracks were invisible. We confirmed Fe 2p narrow scan spectra of the iron oxide on Si(100) and PEN by XPS. Both of the deposited iron oxide Fe 2p peak positions are indicated the film is stoichiometrically close to Fe₂O₃. These suggest the applicability of the present RT-ALD to heat-sensitive materials. To discuss the surface reaction in this process, we picked up a typical mass variation curve, as shown in figure 2. The number of reacted ligands of the precursor adsorption period is estimated as 1 or 2 from a mass variation ratio during one ALD cycle. In the conference, we report the characteristics of deposited iron oxide and discuss the surface reaction of iron oxide RT-ALD.

AF-MoP-8 Simulation of Conformality of ALD Growth Inside Lateral Channels: Comparison between a Diffusion-Reaction Model and a Ballistic Model, J. Velasco, J. Järvilehto, J. Yim, E. Verkama, Riikka Puurunen, Aalto University, Finland

A number of models have been developed for simulating the conformality of atomic layer deposition processes.¹ Simulation models can be used, for example, for the optimization of process parameters towards improved conformality, and for the extraction of kinetic-related information through fitting the model parameters to experimental data.^{2,3}

Three fundamentally different simulation approaches have been described in the literature to simulate the conformality of ALD processes: diffusionreaction, ballistic, and Monte Carlo models.^{1,5,6} In diffusion-reaction models, transport inside a feature is described by a diffusion equation that includes an adsorption loss term.^{1,6} Ballistic models describe particle transport in the molecular flow regime and are based on a balance of particles to compute fluxes at different locations inside a feature.⁵ In Monte Carlo models, the path of each particle at a time is simulated.¹

In this work, we compared the conformality of ALD growth inside lateral channels predicted by two fundamentally different models: a diffusion-reaction model (Model A)^{6,7} and a ballistic model (Model B).^{5,8} The effect of different operating conditions on the conformality of ALD is studied in the free molecular flow regime.

For all the parameters studied, the main trends of the obtained saturation profiles were similar for both models (Fig. 1 of the supp. info). However, penetration depth at half-coverage predicted by the ballistic model was greater than that predicted by the diffusion-reaction model while the predicted slope at half-coverage was greater with the diffusion-reaction

2

model (Fig. 2 of the supp. info). The ballistic model predicted a sudden increment of coverage when the profile reaches the end of the channel. The reasons and consequences of the differences will be discussed.

This work was supported by the Academy of Finland (ALDI consortium,
grant No. 331082). An earlier version of this work was presented at the 21st
Int. Conf. on Atomic Layer Deposition, ALD 2021.

References

1. V. Cremers et al., Appl. Phys. Rev., 2019, 6, 021302.

2. K. Arts et al., J. Vac. Sci. Technol. A, 2019, 37, 030908.

3. J.R. van Ommen, A. Goulas, and R.L. Puurunen. Atomic Layer Deposition. In Kirk-Othmer Encycl. Chem. Technol., John Wiley & Sons, Inc., 2022.

4. A. Yanguas-Gil and J.W. Elam, Chem. Vap. Deposition, 2012, 18, 46.

5. A. Yanguas-Gil and J.W. Elam, Theor. Chem. Acc., 2014, 133, 1465.

6. J. Yim and E. Verkama et al., submitted. Preprint DOI:10.33774/chemrxiv-2021-2j4n1

7. M. Ylilammi, O. Ylivaara, and R.L. Puurunen, J. Appl. Phys., 2018, 123, 205301.

8. Machball software, available at: https://github.com/aldsim/machball (accessed: Jan. 25, 2022).

AF-MoP-9 Chemisorption of Aminosilane Precursors on OH-terminated SiO₂ Surfaces during ALD of SiO₂, *Heeju Son*, *k*. *Khumaini*, *H*. *Roh*, *H*. *Kim*, *W*. *Lee*, Sejong University, Korea (Republic of)

Understanding the surface reaction mechanism is required to design and improve the atomic layer deposition (ALD) process. The ALD reaction mechanism can be studied by simulating the chemisorption of silicon precursor by density functional theory (DFT) calculation, which agrees well with the experimentally observed results [1]. The ALD SiO₂ using an aminosilane-type precursor, such as DIPAS (SiH₃(NⁱPr₂)), BDEAS (SiH₂(NEt₂)₂), or TDMAS (SiH(NMe₂)₃), is being used in various dielectric deposition and patterning processes. DFT studies simulated the chemisorption of the Si precursor, assuming that dialkylamino ligands combine with H atoms of OH groups to form amine molecules as the byproduct. However, there was no report considering the type and distance of the hydroxyl groups on the surface. Also, comparative studies with experimental results are lacking. Therefore, in this study, we compared the chemisorption of aminosilane-type precursors on various OH-terminated SiO₂ surfaces modeled with different hydroxyl types and distances by DFT simulation. We also compared the chemisorption mechanisms of the precursors having different numbers of dialkylamino ligands. Reaction energies were calculated for all possible reaction pathways, assuming the release of amine or H_2 as the byproduct, and then the activation energies were obtained for exothermic pathways. The surface species predicted by DFT simulation well explains the experimental results from in-situ quartz crystal microbalance (QCM) and Fouriertransform infrared (FTIR) spectroscopy.

[1] H. Roh et al., Appl. Surf. Sci. 571 (2022), 151231.

AF-MoP-10 Volatile Adducts of Cobalt and Nickel Halides, *Timo Hatanpää*, A. Vihervaara, M. Ritala, University of Helsinki, Finland

Volatile metal halides in general are good ALD precursors as they are simple, cheap, readily available, thermally stable, reactive with many nonmetal precursors and small in size allowing good growth rates. Unfortunately, halides of many metals are not volatile or they possess only low volatility. Many of such metal halides can, however, be rendered more volatile by adding proper adduct ligands to the metal coordination sphere.

Earlier we have introduced diamine adducts of cobalt and nickel to ALD. While pure halides of Co and Ni have polymeric solid-state structures, high melting point and low volatility, the diamine adducts have monomeric structure and volatility that is among average for known Co and Ni precursors. Under reduced pressure these compounds sublime intact and have sufficiently large temperature window between the sublimation and decomposition temperatures. These compounds have also been shown to perform well in ALD.[1]

For metal films low deposition temperatures are desired to minimize agglomeration and thereby obtain smoother films that are continuous at lower thickness. High volatility of the metal precursor is therefore needed. Here more adducts of the transition metal halides, especially phosphine adducts, were synthesized and studied.

While diphosphine adducts of Co and Ni are thermally exceptionally stable, their volatility is low. In contrast, monophosphine adducts of the same are highly volatile but the thermal stabilities are low. For example NiCl₂(PEt₃)₂ sublimes already at 80 °C under vacuum, but signs of thermal decomposition are also seen already around 130 °C. The temperature window is still large enough to allow ALD usage as demonstrated by Ni metal film deposition using this precursor (another ALD2022 presentation).[2] Monophosphine adducts of Co have lower volatility than the Ni counterparts so that the temperature window between efficient evaporation and thermal decomposition is narrower.

In this presentation the thermal properties of cobalt and nickel halides with different neutral ligands including amines and phosphines are presented and compared. Among other things, it has been found that the thermal stability mainly increases from chlorides to iodides and the trend seems to be more pronounced with cobalt. The increasing molecular weight from chlorides to iodides causes only small decrease in volatility.

[1] K. Väyrynen (2019), Academic Dissertation, University of Helsinki, http://ethesis.helsinki.fi

[2] A. Vihervaara, T. Hatanpää, M. Ritala, ALD2022 abstract (to be published).

AF-MOP-11 Fabrication of Aluminum Nitride with 160 °C Atomic Layer Deposition using Plasma Excited Ammonia, *Kentaro Saito*, *K. Yoshida*, *M. Miura*, *K. Kanomata*, *B. Ahmmad*, *S. Kubota*, *F. Hirose*, Yamagata University, Japan

Aluminum nitride (AIN) has been attracting much attention since it is also expected to be used as a passivation layer for AlGaN/GaN high electron mobility transistors. In the conventional technologies, AIN depositions were examined by atomic layer deposition (ALD), although these were performed at a temperature of 200 °C and higher. To suppress solid-phase diffusion of the passivation layer, a low-temperature deposition is demanded. In this study, the low-temperature ALD of AIN using trimethylaluminum (TMA) and plasma-excited ammonia is examined. We evaluated chemical composition in the AIN films deposited at 160 °C by secondary ion mass spectrometry (SIMS) and cross-sectional transmission electron microscope (cross-sectional TEM), respectively. To confirm its applicability to a gas barrier, we try to deposit AIN on PET.

TMA was used as an Al precursor. Plasma-excited ammonia was used as a nitridation gas. For its generation, 4.2 sccm ammonia and 1.8 sccm argon were mixed, followed by being excited with an RF power of 250 W and a frequency of 13.56 MHz. TMA was introduced with an exposure of 1×10^5 L (1L=1× 10^{-6} Torr \cdot s). The nitridation time was 180 s.

We observed the interface layer to examine if the film was deposited on the substrate without solid phase diffusion. The chemical composition was evaluated to measure impurity in the AIN film. Fig 1. shows a crosssectional TEM image obtained from the AIN with the 150 cycles ALD. We see the polycrystalline AIN film on the starting surface without damage to the Si substrate. To propose the application to a gas barrier, we examined to deposit AIN on PET. Fig. 2 shows an image of an AIN deposited PET sample. From Fig.2, it is considered that the AIN deposition on PET is performed without the thermal deformation of the film. The water barrier characteristics are going to be disclosed in the conference for the further discussion.

AF-MoP-12 On the Underestimation of the Low-Temperature Limit for ALD of MoO_x from Mo(CO)₆ Precursor, *Kees van der Zouw*, *T. Aarnink*, *A. Kovalgin*, University of Twente, Netherlands

Dependencies of the growth rate per cycle (GPC) as a function of the temperature (T), and the so-called precursor saturation curves, are standardly observed for atomic layer deposition (ALD) and reflect its self-limiting nature. Such curves are considered as an indication of the ALD window and required to demonstrate the occurrence of ALD at various circumstances. Plotting GPC versus T is expected to be system independent, meaning that the fundamental features of the GPC curve should be similar between one reactor system and another. It looks feasible to underestimate the limits of the ALD window if there is no universal method to determine the GPC. Based on our experimental findings, we propose an underestimation of the ALD temperature window to occur in the literature study on the (self-limiting) chemisorption of $Mo(CO)_6$ while depositing molybdenum oxide (MoO_x) and molybdenum oxide compounds.

According to the literature, the use of $Mo(CO)_6$ as precursor and O_3 or O_2 -plasma as co-reactant results in the self-limiting chemisorption of $Mo(CO)_6$ roughly between 150 and 175 °C. The GPC is mentioned to quickly decay

for temperatures below 150 °C, with hardly any deposition at T<130 °C [1,2]. According to the experimental observations in this work, the deposition of MoO_x on a silicon substrate can indeed hardly start at T<130 °C, due to the practically inappropriate long incubation time. However, the deposition of MoO_x, once enabled at T=130 °C or higher, naturally continues at 80 °C with a reasonable GPC. The growth per cycle decreases only from 0.028 nm/min at T= 175 °C to 0.020 nm/min at T= 80 °C.

In our presentation, an extended look into these experimental findings including film characterization will be given. We further would like to draw attention of the ALD community to a need for verifying the methodology of ALD window determination for scientific experiments.

[1] J. Mater. Chem., 2011, 21, 705-710

[2] Appl. Mech. Mater., 2014, 492, 375-379

AF-MOP-13 ZnO_{1-x}S_x Layer Growth and Composition by Thermal and Plasma ALD, *Stefanie Spiering*, *D. Bagrowski*, *T. Magorian Friedlmeier*, *W. Hempel*, *T. Wohnhaas*, *J. Becker*, Zentrum für Sonnenenergie- und Wasserstoff-Forschung Baden-Württemberg (ZSW), Germany

ZnO_{1-x}S_x alloys are wide-bandgap materials with high transmittance in the short-wavelength region. This makes them promising candidates as the n-type conducting buffer layer for different absorber types in thin-film photovoltaic devices. In this work, we studied the growth behaviour and composition of ternary ZnO_{1-x}S_x on Si wafer substrates at a fixed deposition temperature of 150°C using a thermal ALD process with diethylzinc (DEZ), H₂O and H₂S as precursors, and additionally a mixed thermal/plasma-enhanced ALD process with DEZ, H₂S and O₂ plasma. The growth rates and the composition of the films were analysed by spectral ellipsometry and EDX (energy dispersive x-ray spectrometry).

With thermal ALD we processed layers over a wide range of different $H_2O:H_2S$ pulse ratios. The cycle ratio for H_2S is defined here as $CR_{H2S}=k^*100\%/(k+l)$ with k being the number of H_2S pulses and l being the number of H_2O pulses in a cycle, respectively. In contrast to former works on $ZnO_{1,x}S_x$ by ALD, not only the smallest possible bilayer period $P=1^*(k+l)$ for a certain CR_{H2S} was applied, but also larger bilayer periods $P=a^*(k+l)$ were examined (e.g. a=1,2,...,7 for $CR_{H2S}=50\%$ (k:l = 1:1) à P=2,4,...,14).EDX measurements showed a significant influence of the bilayer period on the sulphur content in the resulting films. Furthermore, we observed a linear relationship between the resulting S/(S+O) ratio in the film and the number of H_2S pulses for a fixed number of H_2O pulses, and vice versa. Therefore, it is possible to estimate the S/(S+O) ratio for a given pulse ratio by extrapolation. We discuss a few exceptions to the linear behaviour, e.g. for $H_2O:H_2S$ ratio 1:1, 1:1 and k:1.

 $ZnO_{1:x}S_x$ films grown in a mixed thermal-/plasma-enhanced ALD process showed a very different layer growth in comparison to the thermal deposition. With the purely thermal process we deposited layers with S/(S+O) ratios in a wide range from about 10 to 90 % applying moderate $H_2S/(H_2O+H_2S)$ pulse ratios, even a higher S-content in the layers was achieved compared to the theoretically expected one. In contrast, by application of oxygen plasma (O₂PI) as reactant for the ZnO cycle, a very high H₂S pulse fraction was necessary to reach S/(S+O) ratios >30%. With an O₂PI:H₂S ratio of 1:24 (CR_{H2S}=96%) a S/(S+O) ratio of 60 % was measured.

AF-MoP-14 Effect of Precursors on Properties of Atomic Layer Deposited Cr-Ti-O Thin Films, *Mahtab Salari Mehr*, University of Tartu, Institute of Physics, Estonia; *L. Aarik*, University of Tartu, Istitute of Physics, Estonia; *T. Jõgiaas, A. Tarre, H. Mändar*, University of Tartu, Institute of Physics, Estonia

Abstract:

Chromium oxide coatings are technologically important due to their good mechanical, chemical, magnetic, catalytic and optical properties. It has been shown that the properties of Cr₂O₃ coatings can be considerably changed if it is doped with an appropriate element. Chromium containing ternary oxide thin films have previously been grown by chemical vapor deposition or magnetron sputtering methods resulting in hard (hardness 31 GPa) [1]orsuper-hard (40 GPa) coatings [2] or showing good photocatalytic properties [3]. In this work chromium containing ternary compound (Cr-Ti-O) thin films were grown on Si (100) substrates using two different precursor combinations: CrO₂Cl₂ -CH₃OH:TiCl₄-H₂O, and Cr(thd)₃-O₃:TiCl₄- $O_3 by$ atomic layer deposition offering precise control over the concentration of dopant elements in the composition. Film density, roughness, phase composition, refractive index, hardness and Young's modulus were studied in variation of Ti concentration in the thin films. Thin films with average thickness of 115 nm deposited using CrO₂Cl₂-CH₃OH and TiCl₄-H₂Oprecursors exhibited crystalline α -Cr₂O₃ (eskolaite) phase with

density from 4.8 to 5.2 g/cm³. By increasing the number of TiCl₄-H₂O cycles compared to CrO₂Cl₂-CH₃OH cycles, the total intensity of eskolaite reflections decreased. The films with average thickness of 45 nm deposited using Cr(thd)₃-O₃ and TiCl₄-O₃showed crystalline TiO₂ anatase phases with density from 4.0 to 4.7 g/cm³ (Fig. 1). As the growth rate of the Cr₂O₃ deposited using Cr(thd)₃-O₃ (50 pm/cycle) was remarkable lower than that for CrO₂Cl₂-CH₃OH process (70 pm/cycle), the change of the Cr(thd)₃-O₃:TiCl₄-O₃ cycle ratio from 1:30 to 1:1 only decreased the amount of the anatase phase in the film.

References:

[1] Bahrami, Amin, et al. "Structure, mechanical properties and corrosion resistance of amorphous Ti-Cr-O coatings." Surface and Coatings Technology, 374 (2019) 690-699.

[2] Mohammadtaheri, M., et al. "An investigation on synthesis and characterization of superhard Cr-Zr-O coatings." Surface and Coatings Technology, 375 (2019) 694-700.

[3] Chen, Yang, et al. "Synthesis of core-shell nanostructured $Cr_2O_3/C@$ TiO₂ for photocatalytic hydrogen production." Chinese Journal of Catalysis 42 (2021) 225-234.

AF-MoP-15 Ligand Exchange with a Novel Cyclic Alkyl Amido Plumbylene and Transition Metal Complexes, *Kieran Lawford*, *G. Bačić*, Carleton University, Canada; *J. Masuda*, St. Mary's University, Canada; *S. Barry*, Carleton University, Canada

Lead-containing precursors are an active field of study for the deposition of PbS (DOI: 10.1021/acs.chemmater.0c01887). PbO (DOI: 10.1149/1.2789286), and other lead-containing films. During our efforts to find new Pb(II) ALD precursors, we found that the acyclic diamido plumbylene lead(II) bis(N-tert-butyltrimethylsilylamide) (0) undergoes facile thermal decomposition into the homoleptic cyclic (alkyl)(amido) plumbylene (caaPb) bis(N-tert-butyl-2-aza-3,3-dimethyl-1-plumba-3silacyclobutane) (1). Our interest in compound 1 pivoted toward the novel nature of the ligand and the potential to use it to synthesize new ALD precursors. We hypothesised that salt-metathesis between 1 and metal chlorides would produce similar heterocycles (caaM, M=metal). Lead is a good candidate for salt metathesis due to the thermodynamic driving force of the formation and precipitation of PbCl₂, driving the reaction to completion. Additionally, the high yield of the caaPb, cheap starting materials, and recyclability of the PbCl₂ by-product makes this synthetic methodology appealing, economical, and green. We tested the metathesis with ZnCl₂ because, if the salt metathesis is possible with zinc, then less electropositive metals should also undergo in the ligand exchange. During our test, we isolated PbCl₂ indicating that the metathesis was successful. Quantum chemical studies suggests the analogous reaction with other metal chlorides should also proceed.

There are early examples of titanium, zirconium, and hafnium complexes with a similar heterocyclic system $(CH_2Si(Me_2)NSiMe_3)$, and the heterocyclic $Cp_2TiCH_2Si(Me_2)NSiMe_3$. was even used as a titanium-ceramic CVD precursor.^{1,2} However, the synthesis of these is unreliable.² We can now prepare similar metal complexes using ligand exchange from **1** with higher yields. Additionally, we expect to see a change in thermal properties after exchanging the trimethylsilyl group for a t-butyl group appended to the heterocycle, since the trimethylsilyl moiety is thermally active. A variety of complexes will be vetted to determine which work best as vapour-phase precursors with respect to volatility and thermal stability.

1. Planalp, R. P.; Anderson, R. A.; Zalkin, A. Dialkyl Bis[bist(trimethylsilyl)amido] Group 4A Metal Complexes. Preparation of Bridging Carbene Complexes by γ Elimination of Alkane. Crystal Structure of {ZrCHSi(Me)₂NSiMe₃[S(SiMe₃)₂]}₂. Organometallics**1983** 2, 16-20.

2. Simpson, S. K.; Anderson, R. A. Reaction of the Metallocene Dichlorides of Titanium(IV) and Zirconium(IV) with Lithium Bis(trimethylsilyl)amide. *Inorg. Chem.* **1981**, *20*, 3627-3629.

AF-MoP-16 Diaminosilane Precursors for the Atomic Layer Deposition of Iron Metal, *Molly Alderman, S. Barry,* Carleton University, Canada

Currently there are few processes for the atomic layer deposition of iron metal thin films, and none without significant limitations—either by CVD or relying on very specific substrates with growth terminating after the initial substrate is covered. With an appropriate precursor, iron films could provide the basis for ferromagnetic coatings and to enable a range of iron alloy films – like stainless steel thin films – with desirable properties.

Geminal diaminosilane (gDAS) ligands are N,N' κ^2 chelates with silicon bridging the coordinating nitrogen centres (Figure 1). We have previously reported a range of first row transition metal complexes with the gemdiaminosilane (^tBuNH)SiMe₂NMe₂ [2]. Transition metal gDAS precursors possess better volatility and thermal stability than their amidinate analogues while still avoiding problematic metal-oxygen bonds (compared to alkoxides) and with added stability from the chelate effect.

Here we report our initial study of new iron complexes which improve on the previously reported acetamidinate and α -imino alkoxide precursors [1]. We have synthesized, characterized, and evaluated the thermal characteristics of several Fe(II) and Fe(III) complexes containing monoanionic and dianionic gDAS ligands which show promise as precursors for the ALD of iron. Preliminary deposition studies using the previously reported Fe(gDAS)₂ monitored by quartz-crystal microbalance, as well as microscopy and initial compositional analysis will also be discussed.

[1] Kalutarage, L. C.; Martin, P. D.; Heeg, M. J.; Winter, C. H. Volatile and Thermally Stable Mid to Late Transition Metal Complexes Containing α -Imino Alkoxide Ligands, a New Strongly Reducing Coreagent, and Thermal Atomic Layer Deposition of Ni, Co, Fe, and Cr Metal Films. J. Am. Chem. Soc.2013, 135 (34), 12588-12591. https://doi.org/10.1021/ja407014w.

[2] Griffiths, M. B. E.; Zanders, D.; Land, M. A.; Masuda, J. D.; Devi, A.; Barry, S. T. (TBuN)SiMe2NMe2-A New N,N'-K2-Monoanionic Ligand for Atomic Layer Deposition Precursors. Journal of Vacuum Science & Technology A2021, 39 (3), 032409. https://doi.org/10.1116/6.0000795

AF-MoP-17 Low-temperature Epitaxy of β-Ga₂O₃ Films via Hollow-cathode Plasma-assisted ALD, Saidjafarzoda Ilhom, A. Mohammad, D. Shukla, B. Willis, University of Connecticut; A. Okyay, Stanford University; N. Biyikli, University of Connecticut

Gallium oxide (Ga₂O₃) is an emerging ultrawide-bandgap (UWBG) semiconductor attracting significant interest for high-power and highfrequency electronics that features comparable electrical properties (larger bandgap ~4.9 eV, high dielectric constant 6-8 MV/cm) to wide-bandgap GaN and SiC. However, growing device-level high-quality (Ga₂O₃) films have been mainly possible at high substrate temperatures (700 - 1000 °C) using complex reactor settings, which substantially increases the production cost and limits the application space.

In this work, we report on low-temperature as-grown crystalline β -Ga₂O₃ films on Si, glass, and sapphire via hollow-cathode plasma-assisted atomic layer deposition (HCPA-ALD). The films were deposited using triethylgallium (TEG) and Ar/O₂ plasma as metal precursor and oxygen coreactant, respectively. Growth experiments have been performed at 240 °C substrate temperatures and 50 W rf-power level. Additionally, eachunit ALD-cycle was followed by an in-situ Ar-plasma annealing treatment, which consisted of Ar-plasma exposure for 20 seconds at 250 W rf-power. Both in-situ and ex-situ ellipsometry were employed to measure the thickness and optical properties of the films. The samples exhibited enhanced growth-per-cycle (GPC) values around 1.3 Å. X-ray diffraction (XRD) of the sample on sapphire revealed epitaxial Ga₂O₃ film signature with monoclinic β -phase. On the other hand, GIXRD of the samples grown on Si and glass displayed polycrystalline β-Ga₂O₃ films. High-resolution transmission electron microscopy (HR-TEM) revealed the epitaxial relationship of the Ga₂O₃ layers grown on sapphire substrates. X-ray photoelectron spectroscopy (XPS) measurements shed light on the elemental composition and provide additional understanding on the nature of chemical bonding states within β -Ga₂O₃ layers. Our results demonstrate the effectiveness of in situ Ar-plasma annealing process within a customized HCPA-ALD reactor setting to transform amorphous wide bandgap oxide semiconductors into epitaxial films at substantially reduced substrate temperatures.

AF-MoP-18 Higher Effective Dielectric Constant of Hafnium Oxide When Grown with Hydrogen Peroxide Compared to Water Vapor, Jeffrey Spiegelman, RASIRC; H. Kashyap, A. Kummel, University of California at San Diego

As high-k gate dielectrics continue scaling in the sub 1nm EOT range, the low-k interfacial layer that is pre-formed or formed during the dielectric deposition process is increasingly problematic since it reduces the effective k-value of the dielectric layer(1) . For Hafnium Oxide grown on silicon, the EOT is a combination of the silicon dioxide interface and the quality of the hafnium oxide. Consiglio (1) reported reduced thickness of the silicon dioxide interface layer with hydrogen peroxide gas when compared to ozone. In this study hafnium oxide was grown on HF cleaned silicon with either water vapor or H2O2 gas. The Keffective was improved by 27% H2O2 grown HfO2. The film grew 50% faster with H2O2 than water. Vt was 0.18V for H2O2 and -1.3V for water. Additional data will be presented to clarify if Monday Afternoon, June 27, 2022

the Keffective was due to a thinner SiO2 interface layer or improved HfO2 film quality.

AF-MoP-19 Atomic Layer Deposition of Low-resistivity Molybdenum Nitride Using F-free Mo Precursors and NH₃ Plasma, Wangu Kang, J. Ahn, J. Han, Seoul National University of Science and Technology, Korea (Republic of)

Molybdenum nitride film (MoN_x) is considered as a promising material in CMOS manufacturing process as gate electrode, contact layer, and diffusion barrier due to its high electrical conductivity, chemical stability, and thermal stability. With the continuous miniaturization of semiconductor devices, the development of atomic layer deposition (ALD) MoN_x process has been demanded to obtain conformal MoN_x film in complex 3-dimensional structure. However, MoN_x ALD using fluorine(F)containing Mo precursor such as MoF₆showed various issues for example, forming void due to HF fume and degradation of metal or peripheral oxide HF by-product. bv In this study, we have developed the ALD MoN_x films using F-free Mo precursor and NH₃ plasma at the temperature range of 200-400 °C. The characteristics of MoN_x thin films such as crystallinity, chemical binding state, impurity and resistivity were examined by GAXRD, XPS, and 4-point measurement system. The post-annealing by rapid thermal annealing (RTA) at 500-800°C was conducted in H₂/Ar ambient to improve the resistivity of MoN_x films, and consequently the significant decrease in resistivity was observed after RTA due to the reduction of MoN_x into Mo.

AF-MoP-20 Atomic Layer Deposition of Molybdenum Carbides for Advanced Metallization: The Influences of Mo Precursor and Post-Deposition Annealing, Ji Sang Ahn, W. Kang, J. Han, Seoul National University of Science and Technology, Korea (Republic of)

As the width of metallization wire in semiconductor device decreases, Cu interconnect and W word line does not scale down as fast as linewidth due to its high electron mean free path (EMFP). Meanwhile, molybdenum (Mo) and molybdenum carbides (MoC_x) are considered as promising materials for next-generation interconnect owing to its small EMFP andless resistivity size effect.In order to apply the Mo and MoCxthin films to the 3dimensional semiconductor devices, atomic layer deposition (ALD) of Mo and MoC_x has been required. However, Mo and MoC_x ALD using fluorine (F)-containing Mo precursor resulted in various issues such as oxide damage and deterioration of device property due to F contamination and HF by-product. Therefore, it is essential to developMo and MoCx ALD processusing F-free Mo precursors.

In this study, δ -MoC and β -Mo₂C thin films were deposited by ALD using two different F-free precursors. Rapid thermal annealing (RTA) process was adopted to reduce MoC_x into Mo metal and to increase crystallinity of the film. β -Mo₂C showed improved crystallinity after RTA process, whereas δ -MoC was reduced into metallic Mo (~46.5 $\mu\Omega$ ·cm) after post-reduction annealing. The crystallinity, chemical state, impurity, and electrical characteristics of molybdenum carbide thin films will be compared before and after annealing process and reaction mechanism of the post-reduction annealing will be discussed.

AF-MoP-21 ALD of Niobium Oxide (Nb₂O₅) and Niobium-Doped Titanium Oxide (Nb:TiO2) for Solar Cell Applications, Thomas Vincent, D. Coutancier, P. Dally, IPVF (Institut Photovoltaïque d'Île-de-France), France; M. Bouttemy, M. Fréqnaux, ILV (Institut Lavoisier de Versailles) - IPVF (Institut Photovoltaïque d'Île-de-France), France; N. Schneider, CNRS-IPVF, France

Atomic Layer Deposition (ALD) is increasingly contributing to the energy field and more specifically to the engineering of solar cells. Its conformity enables deposition on nanostructured substrates and its low growth temperature allows the deposition on temperature-sensitive substrates such as perovskite.

Niobium oxide, Nb₂O₅, is a wide bandgap semiconductor that has been grown by different methods and has recently been used in solar cells. Its optical and electrical properties depend strongly of the technique used for its growth, opening access to a wide range of application, such as electron transport layer (ETL)or passivation layer [1,2]. It is also used for the doping of titanium oxide (TiO₂), a well-known ETL, to reach a better stability of the complete solar cell.

In this work, we first developed ALD-Nb₂O₅ from tris(diethylamido)(tertbutylimido)niobium (TBTDENb) and water. The growth was studied from 100°C to 200°C, on several substrates (Si, glass, FTO, ITO). The ALD process was first optimized using QCM (Quartz Crystal Microbalance). Annealing studies in different conditions (air/inert atmospheres, up to 600°C) were

5

conducted to understand the evolution of Nb_2O_5 at temperatures which are relevant within the fabrication steps of the solar cell.

Then, niobium-doped titanium oxide was developed using tris(isopropoxide)titanium (TTIP) as titanium precursor, and Nb-doping was applied by a supercycle strategy and tuned by several methods (supercycle ratio, precursors sequences). For both materials, structural, chemical, electrical and optical properties were characterized by XRR (X-Ray Reflectivity), GIXRD (Grazing Incidence X-Ray Diffraction), ellipsometry, SEM (Scanning Electron Microscopy), XPS (X-Ray Photoelectron Spectrometry), XRF (X-Ray Fluorescence), spectrophotometry, 4-points probe.

Finally, those ALD-materials were implemented in perovskite solar cell architectures.

[1] Subbiah, et al (2019). *Energy Technology, 8*(4), 1900878. https://doi.org/10.1002/ente.201900878

[2] Macco et al, (2018). *Solar Energy Materials and Solar Cells, 184*, 98-104. https://doi.org/j.solmat.2018.04.037

AF-MOP-22 Understanding the Grown Mechanism of Al₂O₃ Thin Films Using Plasma-Activated Deionized Water as Oxygen Source on Thermal ALD, William Chiappim, Aeronautical Institute of Technology, Brazil; J. Chaves, Aeronautical Institute of Technology, Brazil; B. Botan Neto, L. Gonçalves de Lima, M. Shiotani Marcondes, N. Galvão, A. da Silva Sobrinho, R. Sávio Pessoa, Aeronautical Institute of Technology, Brazil

The atomic layer deposition (ALD) of metallic oxides, mainly alumina (Al2O3), when performed in thermal mode, uses deionized water (DI) as an oxygen source and trimethylaluminum (TMA) as a metal reactant. However, growth per cycle (GPC) of Al2O3 thin films for the reactant and co-reactant mentioned above is limited to 0.1 nm/cycle. This barrier in the GPC is overcome by using plasma technology as an oxygen source. This technique is commonly called energy-enhanced ALD because the plasma oxygen source provides tremendous activation energy during the coreactant step, which allows for greater efficiency in generating active sites on the substrate surface, promoting thus more reactions between the surface and the metal reactant. This process, called plasma-enhanced ALD (PEALD), is commonly used to replace DI water with O2 plasma as an oxygen source [2]. It is reported in the literature that for the TMA reactant, this replacement of the vapor phase oxygen source (thermal ALD) by a plasma oxygen source (PEALD) generates an increase in the alumina GPC to 0.12 nm/cycle, i.e., a gain of 20%. However, this gain in the GPC has a high cost, as the PEALD uses a source to generate the plasma that uses electrical energy to ionize and dissociate the gas introduced into the chamber. In addition, it is necessary to design a new deposition chamber and design the entire electrical system to turn it into equipment that operates in plasma mode, which significantly increases production costs. The present work presents a cheap alternative to increase the GPC of alumina by 20%. A gliding arc plasma and compressed air were used to activate DI water. Plasma-activated water (PAW) was prepared by a forward vortex flow reactor (FVFR) type with air compressed at atmospheric pressure. The activation times were 10, 30, and 60 min, and it was reached the following pH of 3.5, 3.0, and 2.5. PAWs were characterized by UV-vis spectrophotometer and presented reactive oxygen and nitrogen species (RONS), namely, H2O2, HNO2, NO2-, and NO3-.

After activation, plasma-activated water (PAW) is carried out into a recipient and introduced in the line of oxygen source in thermal ALD. The ALD pulse times were 0.15-30-0.3-30 s, TMA, N2 purge, PAW, and another N2 purge. The number of cycles was fixed at 1000 cycles, and the substrate was the Si (100). Alumina thin films growth was characterized *in-situ* by mass spectrometry and *ex-situ* by an optical profilometer, FT-IR and FEG-SEM.

According to the characterizations mentioned above, the existing RONS in PAW probably contributed to the activation of sites in the Si(100) substrate, thus increasing the GPC of the alumina.

AF-MOP-23 Atomic Layer Deposition of Titanium Phosphate onto Reinforcing Fibers Using Titanium Chloride, Water, and Tris-(Trimethylsilyl)-Phosphate as Precursor, Pauline Dill, X. Ren, H. Hintersatz, University of Technology Chemnitz, Germany; M. Franz, Fraunhofer Institute of Electronic Nano Systems ENAS, Germany; D. Dentel, C. Tegenkamp, S. Ebert, University of Technology Chemnitz, Germany

Thermal ALD process for depositing titanium phosphate on carbon fiber bundles and flat silicon substrates using the novel precursor system titanium tetrachloride, water, and tris-(trimethylsilyl)-phosphate was studied. We observe a linear growth per cycle of 0.22 nm/cycle. Compositional analysis of the coating was investigated using energydispersive X-ray spectroscopy (EDXS), and X-ray photoelectron spectroscopy (XPS). The thermal behavior up to 900 °C and the changes compositional were investigated via infrared spectroscopy (IR). With oxidation of coated carbon fibers in the thermogravimetric analysis (TGA) we could show that the coating also increases the onset temperature of oxidation of the fibers.

The coated fibers will be embedded in a ceramic matrix to give a fiber reinforced ceramic matrix composite (CMC). In this CMC the coating has the role induce the deflection of cracks that run through the matrix and to provide oxidation resistance.

P. Dill, X. Ren, H. Hintersatz, M. Franz, D. Dentel, C. Tegenkamp, S. Ebert, *J.Vac. Sci. Technol. A* 40 (**2022**); doi: 10.1116/6.0001514

AF-MoP-24 Amorphous Carbon(a-C) Atomic Layer Deposition using CBr₄ Precursor, *Tae Hyun Kim*, *M. Kim*, *S. Park*, *S. Chung*, *H. Kim*, Yonsei University, Korea

Phase change memory (PCM) is attracting attention as the need for highperformance memory semiconductors that can process large amounts of data quickly increases with the development of the 4th industry, such as big data and artificial intelligence. At the same time, research on 3dimensions cross-point (3D X-point) memory using memory cells and selectors in the area where word lines and bit lines intersect in 3D to create a highly integrated PCM is also being conducted. However, the most important factor for high efficiency and integration of these PCM devices is to reduce the current required for device operation. Recently, studies have been actively conducted to increase the joule heating efficiency by increasing the specific resistance of the electrode. In this respect, carbon, which exhibits various resistivities (conductor (graphite, sp² bond) insulator (diamond, sp³ bond)) ^{[1], [2]} depending on the state of atomic bonding, is an attractive material as a PCM electrode. So far, PVD (Physical Vapor Deposition) and CVD (Chemical Vapor Deposition) processes have been used to deposit carbon thin films. However, it is difficult to deposit a uniform carbon thin film on a three-dimensional structure with the conventional method, and it is difficult to control the physical properties of carbon. Therefore, in this study, a carbon process using ALD (Atomic Layer Deposition) was developed. Furthermore, the analysis of the carbon thin film properties according to the process conditions (reactant, temperature, pressure, time, etc...) was analyzed, and the corresponding mechanism was studied.

AF-MoP-25 Al₂O₃ Nanowire Growth on WSe₂ Flake by Using Lowtemperature ALD, *Chi-Chung Kei*, *C. Su*, *B. Liu*, Taiwan Instrument Technology Institute, National Applied Research Laboratories, Taiwan; *Y. Su*, Department of Electrophysics, National Yang Ming Chiao Tung University, Taiwan; *T. Lee, C. Cheng, H. Wang, I. Radu*, Corporate Research, TSMC, Taiwan; *W. Chang*, Department of Electrophysics, National Yang Ming Chiao Tung University, Taiwan

Transition metal dichalcogenide (TMD) has been considered a promising material for the channel material of electronics due to remarkable electrical characteristics. Atomic layer deposition (ALD) is a promising method to deposit film for TMD-based device fabrication. However, lack of dangling bond on TMDs leads to a chemically inert surface for the chemisorption of ALD precursors. Physisorption of ALD precursors is considered to form seeding or reactive sites on chemically inert surface of TMDs. In this work, WSe₂ flakes were used as the substrate to grow Al₂O₃ by using ALD. 90°C was chosen as the working temperature to allow the physisorption of ALD precursors on WSe₂ substrates. Atomic force microscopic image (AFM) in Fig. 1(a) shows the triangular WSe₂ flakes were grown on mis-cut sapphire substrates after meta-organic vapor deposition. WSe₂ surface, shown in Figs. 1(b) and (c), exhibits terrace morphologies and is similar to mis-cut sapphire substrate. Terrace width and step-height are estimated to about 80 nm and 0.3 nm, respectively. In the Al₂O₃ ALD, trimethyl aluminum (TMA) and water were used as the sources of aluminum and oxygen, respectively. One ALD cycle for growing Al₂O₃ started with 0.1 s TMA pulse, followed by 18 s N2 purge to remove excess precursor and by- products, then 0.1 s water pulse, and finally another 18 s N2 purge. After 20 cycles of Al2O3 ALD, nanowire-like structure was formed on the surface of WSe₂ flake, as shown in Fig. 2(a). According to the linescan data in Figs. 2(b) and (c), the width and height of nanowires are estimated to be about 40 and 6 nm, respectively. The peak-to-peak spacing between nanowires, about 80nm, is consistent with the terrace width of WSe₂ surface. This suggests that Al₂O₃ preferentially nucleates and grows along the valley of terrace structure. This can be attributed that the energy barrier to form a stable physisorbed embryos on a concave valley is lower

than that on a flat surface. The growth per cycle (GPC) of about 0.3 nm/cycle also suggests that Al_2O_3 growth was both contributed from the physisorption and chemisorption of ALD precursors.

AF-MoP-26 Comparison of Atomic Layer Deposited ZrO2 Thin Film using Cp-based Zr Precursor and Newly Synthesized Cp-based Zr Precursor, *Hyemi Han, S. Park, S. Na, S. Chung, H. Kim,* Yonsei University, Korea (Republic of)

As the dynamic random access memory (DRAM) has been continuous downscaling, zirconium oxide, a high-k dielectric material, is widely used as a substitute for conventional SiO2. Because it has a relative high dielectric constant (15~22), a high breakdown field (15–20 MV/cm), a large band gap (5–7 eV), and a good thermodynamic stability up to 800 °C when in contact with the silicon substrate.¹

Among the various methods of deposition, ALD has been recognized as the leading candidate to process the high-k dielectrics owing to characteristics such as self-limiting reaction, excellent conformality, easy controlled thickness, and large area uniformity.²

The choice of precursor is very important because the film properties are totally different depending on which kinds of precursor is used. For instance, in case of halide ligand precursor, it has been widely used since there is no carbon incorporation into the films. However there is limitation like low volatility, particle issue and harmful byproduct. Also alkoxide ligand having a strong metal-oxygen bonding is required a relatively high temperature and it is easily decomposed by heat and contains high concentration of carbon impurity in the thin film.³ Among various types of precursors, ZrO2 thin film using Cp ligand precursor have been widely used recently due to their good thermally stable allowing deposition temperature above 300°C and excellent crystallinity. However it has disadvantage in that the growth rate is decreased due to its bulky size.⁴

Like this, several kinds of precursors have been studied for ALD of ZrO2 thin film but it is still needed for research on precursors. In this paper, new linked-Cp zirconium precursor ((linked CpZr(N(CH3)2)3)) was synthesized for ALD of ZrO2 thin film and compared with the commonly used (CpZr(N(CH3)2)3) precursor in terms of growth characteristic and film property. For ALD ZrO2 process, oxygen plasma was used as the oxidant.

Various experimental methods, including spectroscopic ellipsometry, X-ray photoelectron spectroscopy (XPS), X-ray diffraction (XRD), X-ray Reflectivity (XRR) and density functional theory calculations (DFT)) were used for analyzing growth characteristic and film property. Additionally for the analysis electrical property, capacitance-voltage (C-V) and current-voltage (I-V) measurements were conducted.

¹ M. Balog, M. Schieber, Thin Solid Films 1977, 47, 109

² H. J. Kim, H. -B. -R. Lee, Thin Solid Films 517, 2563–2580 (2009)

³ J. Niinistö, K. Kukli, Advanced Engineering Materials vol. 11 223–234 (2009)

⁴ J. –S. Jung, S. –K. Lee, Thin Solid Films. Thin Solid Films.Volume 589 831-837 (2015)

AF-MoP-27 Thermal Atomic Layer Deposition of in_2O_3 Thin Films Using a Homoleptic Indium Triazenide Precursor and Water, *Pamburayi Mpofu*, Linköping University, Sweden

Indium oxide (In₂O₃) has received much attention for a wide range of applications, including in optoelectronics as a transparent conducting material. We report ALD of In2O3 using a recently reported indium(III) triazenideprecursor¹ together with water. The deposition process was studied between 150 and 520 $^\circ\mathrm{C}$ using a homebuilt crossflow ALD reactor at 50 hPa with indium(III) triazenide pulsed for 4s, and water pulsed for 3s as precursors and purging with nitrogen for 10s in between. The deposition process is self-limiting at ~1.0 Å/cycle and temperature window between 270 and 385 °C. XRD analysis shows that the films are polycrystalline with a preferred (222) orientation. The films are substoichiometric with have low levels of C impurities. Optical transmittance is high, >70% in visible light, and the resistivity was found to be low signifying high conductivity. These results are on par with the current state-of-the-art reported for thermal ALD of In₂O₃ from a formamidinate precursor.² We make a direct comparison between the indium triazenide precursor and the indium formamidinate precursor in the same reactor and show that they render very similar temperature windows and film quality optical transparency, and conductivity of the deposited films.

Refs.:

O'Brien et al. Chem. Mater., 2020, 32, 4481.
Kim et al. Chem. Eur. J., 2018, 24, 9525.

AF-MoP-28 Nanoindentation of Amorphous Aluminium and Tantalum Oxide Nanolaminates, *Helle-Mai Piirsoo*, *T. Jõgiaas, P. Ritslaid, K. Kukli, A. Tamm*, University of Tartu, Institute of Physics, Estonia

Ta₂O₅-Al₂O₃ nanolaminates are potentially applicable as high dielectric strength insulators [1], resistive switching media [2] and corrosion resistant coatings [3]. For most applications the mechanical properties of the thin films have a significance on the reliability of the devices. Double- and triple-layer amorphous Al₂O₃-Ta₂O₅ laminates with an overall thickness of about 70 nm were atomic layer deposited while changing the sequence of the layers from surface to substrate. Hardness and elastic modulus of the laminates were measured with nanoindentation and influence from the layer thickness and sequence on the mechanical properties were analysed.

Figure 1 depicts the change of hardness along the depth of the films for double-layered laminates. Layered structure caused an uneven rise in hardness with depth dependent on the sequence of Al_2O_3 and Ta_2O_5 . Figure 2 describes the hardness of triple-layered laminates, showing an even incline for the laminate with a middle Ta_2O_5 layer surrounded with Al_2O_3 whereas the laminate with middle Al_2O_3 between Ta_2O_5 layers resembled the double-layered laminate behaviour. An additional quadruple-layered laminate was measured and showed a steady increase of hardness with depth. The elastic modulus for all the laminates was similar, steady along the depth and fell between 145 - 155 GPa.

It can be concluded that the hardness of layered films is affected by both the thickness of the layers and their sequence. Lowering a single constituent layer thickness can smoothen the difference in hardness along the film depth which could reduce internal stresses, defects, and delamination. Adding an extra layer to a thin film could increase its mechanical resilience in a cost-effective way.

References:

[1] B.V.T. Hanby, B.W. Stuart, M. Gimeno-Fabra, J. Moffat, C. Gerada, D.M. Grant, Appl. Surf. Sci. 492, 328 (2019) https://doi.org/10.1016/j.apsusc.2019.06.202.

[2]W. Song, W. Wang, H.K. Lee, M. Li, V.Y.-Q. Zhuo, Z. Chen, K.J. Chui, J.-C. Liu, I.-T. Wang, Y. Zhu, N. SinghAppl. Phys. Lett. 115, 133501 (2019) https://doi.org/10.1063/1.5100075

[3] B. Díaz, E. Härkönen, J. Swiatowska, A., V. Maurice, M. Ritala, P. Marcus, Corr. Sci 82, 208 (2014) http://dx.doi.org/10.1016/j.corsci.2014.01.024

AF-MoP-29 Experimental and ReaxFF MD Studies for Boron Nitride ALD Growth from BCl₃ and NH₃ Precursors, *Naoya Uene*, *T. Mabuchi*, Tohoku University, Japan; *M. Zaitsu*, Japan Advanced Chemicals Ltd., Japan; *Y. Jin*, Japan Advanced Chemicals Ltd., Japan, China; *S. Yasuhara*, Japan Advanced Chemicals Ltd., Japan; *A. van Duin*, Pennsylvania State University, USA; *T. Tokumasu*, Tohoku University, Japan

Boron nitride (BN) has been considered a promising dielectric material for 2D material-based electronics. The atomic layer deposition (ALD) technique is a good method to grow conformal and ultrathin materials at relatively low temperatures. However, the growth mechanism is still not clear because the surface events with self-termination resulting from chemical reactions and physical dynamics are complicated. We aim to understand the growth mechanisms of the BN-ALD process from BCl₃ and NH₃ by experiments and simulations using the reactive force-field molecular dynamics (ReaxFF MD).

First, we investigated temperature profiles using a thermal ALD system described in Fig. (a). BN films were grown on the Si(100) surface in a hot walled horizontal reactor through the general ALD cycles as follows; BCl₃ exposure, Ar purge, NH₃ exposure, and Ar purge. Fig. (b) shows the thickness of the grown BN as a function of growth temperature measured on the chamber wall. We obtained a relatively small dependency of growth per cycle on the temperature in the range of 700-900 °C, suggesting ALD growth. The thin film property was characterized using IR spectroscopy as shown in Fig. (c). The remarkable peak at 1367 cm⁻¹ is originated from sp² BN associated with the in-plane stretching. These experimental results mean proper BN growth on the Si(100) surface by ALD. Second, we started to develop a new force field for the ReaxFF MD, which can simulate the surface events including complicated chemical reactions and physical dynamics. The initial force field is based on the two types of force field; boron nitride nanostructure formation and dynamical crack propagation in silicon, by Lele et al. and Buehler et al., respectively. The initial force field is mainly trained for boron chloride species and the reaction of BCl3 on the

OH-terminated Si(100) surface. We simulated one ALD cycle after thermal annealing to relax the initial system shown in Fig. (d). The Ar purge steps are simply replaced as the removals of gas molecules with thermal annealing. The simulations revealed some growth mechanisms: the $-BCl_2$ chemisorbed on the OH-terminated surface in the feed step, $-NH_2$ chemisorbed on the CI-terminated surface in the reaction step, and HCI was generated with the chemisorption in these two steps. Our simulations are still one ALD cycle, however, these growth mechanisms are chemically straightforward.

We will show you thin film properties such as crystallinity and composition and compare them with the experimental results. The experimental and theoretical study of the ALD can be applied not only to the BN system but also to critical materials such as TiN and GaN in the future.

AF-MOP-30 Enhancement of Graphene-Related and Substrate-Related Raman Modes Through Dielectric Layer Deposition, Karolina Pietak, Warsaw University of Technology Faculty of Chemistry, Poland; J. Jagiełło, A. Dobrowolski, University of Warsaw Faculty of Physics, Poland; R. Budzich, Warsaw University of Technology Faculty of Chemistry, Poland; A. Wysmołek, University of Warsaw Faculty of Physics, Poland; T. Ciuk, Łukasiewicz Research Network - Institute of Microelectronics and Photonics, Poland

Enhancement of the Raman signal intensity is currently among the most researched directions in developing Raman-based characterization techniques of all 2D materials as it elevates detection limits of their fine structural properties. The interest in signal intensification is also triggered by the wide range of applications it can benefit, including biochemistry and biosensing, polymer and materials science, catalysis, electrochemistry, the study of high-temperature processes, and detection of hazardous gases.

In this research, we demonstrate a method for the enhancement of Raman active modes in hydrogen-intercalated quasi-free-standing epitaxial Chemical Vapor Deposition graphene and the underlying semi-insulating 6H-SiC(0001) substrate through constructive signal interference within the atomic-layer-deposited amorphous Al_2O_3 passivation. We find that an optimum Al_2O_3 thickness of 85 nm for the graphene 2D mode and 82 nm for the SiC longitudinal optical A_1 mode at 964 cm⁻¹ enables a 60% increase in their spectra intensities. We demonstrate the method's efficiency in Raman-based determination of the dielectric thickness and high-resolution topographic imaging of graphene surface [1].

[1] K. Piętak, J. Jagiełło, A. Dobrowolski, R. Budzich, A. Wysmołek, T. Ciuk, Enhancement of graphene-related and substrate-related Raman modes through dielectric layer deposition, accepted for publication in *Applied Physics Letters* (28.01.2022).

Acknowledgements:

The research leading to these results has received funding from the Research Foundation Flanders (FWO) under Grant No. EOS 30467715, the National Science Centre under Grant Agreement No. OPUS 2019/33/B/ST3/02677 for project "Influence of the silicon carbide and the dielectric passivation defect structure on high-temperature electrical properties of epitaxial graphene" and the National Centre for Research and Development under Grant Agreement No. LIDER 0168/L-8/2016 for the project "Graphene on silicon carbide devices for magnetic field detection in extreme temperature conditions".

Karolina Piętak and Rafał Budzich acknowledge financial support from theIDUBproject(ScholarshipPlusprogramme).

AF-MoP-31 Interface Modification of Thermoelectric Materials with Oxide Compounds by Atomic Layer Deposition, *Shiyang He, A. Bahrami, K. Nielsch,* Leibniz-Institut für Festkörper- und Werkstoffforschung Dresden e.V., Germany

In thermoelectric materials, phase boundaries are crucial for carrier/phonon transport. Manipulation of carrier and phonon scatterings by introducing continuous interface modification has been shown to improve thermoelectric performance. In this work, a strategy of interface modification based on powder atomic layer deposition (PALD) is introduced to accurately control and modify the phase boundary of pure bismuth. Ultrathin layers of Al₂O₃, TiO₂, ZnO and Sb₂O_x are deposited on Bi powder

by typically 1–20 cycles. All of the oxide layers significantly alter the microstructure and suppressed grain growth. These hierarchical interface modifications aid in the formation of an energy barrier by the oxide layer, resulting in a substantial increase in the Seebeck coefficient that is superior to that of most pure polycrystalline metals. Conversely, taking advantage of the strong electron and phonon scattering, an exceptionally large decrease in thermal conductivity is obtained. It's worth noting that a substantial decrease of κ_{tot} from 7.8 to 5.7 W·m⁻¹·K⁻¹ was obtained with just 5 cycles of Sb₂O_x layers and a 16% reduction of κ_{lat} . Finally, a maximum figure of merit, *zT*, of 0.15 at 393 K and an average *zT* of 0.14 at 300–453 K were achieved after 5 cycles of Al₂O₃-coated Bi. The ALD-based approach, as a practical interfacial modification technique, can be easily applied to other thermoelectric materials to enhance their performance.

AF-MoP-32 Comparative Study of the Growth Characteristics and Electrical Properties for Atomic Layer Deposited HfO₂ Films Using Cp-Based Hf Precursors, Seunggyu Na, S. Park, H. Yoon, Yonsei University, Korea; Y. Lee, Stanford University; S. Chung, H. Kim, Yonsei University, Korea

Scaling down of Si-based metal-oxide-semiconductor (MOS) has been main issue for semiconductor industry. With the demand for faster and smaller devices, channel length and thickness of gate dielectric have rapidly shortened. To prevent high leakage current for thin SiO₂, scaling down requires materials with higher dielectric constant. Among many materials, HfO₂ has been widely used for its superior properties. It has appropriate band offset with Si (~1.4 eV) and superb dielectric properties, such as high dielectric constant of about 25. For deposition of gate dielectric HfO₂, atomic layer deposition (ALD) technique has been widely used due to its superior characteristics, including excellent conformality, easily controlled thickness at atomic scale, atomic level composition control, large area low impurity, and low uniformity. growth temperature.

Many kinds of precursors, such as halides, alkoxides, alkylamides, and cyclopentadienyls have been studied for HfO_2 ALD process. Among these various type of precursors, alkylamide precursors are attracting attention due to their superior characteristics. They have relatively weak metal-N bonds, and weaker bond attributes to high reactivity in low temperature. Furthermore, the ligands of alkylamide precursors effectively prevent neighboring precursors from bonding to the metal center, which attributes to low melting point and high volatility. However, their weak metal-N bonds lead to decomposition of the precursors at high temperature. It limits ALD window and results in incorporation of impurities in film at high process temperature. In addition, the upper process temperature limit also prevents the formation of films with higher density and higher crystallinity.

To overcome this low thermal stability issue, heteroleptic precursors including cyclopentadienyl (Cp) ligand were introduced. Cp ligand has been widely employed for enhancing thermal stability and volatility of precursor. To take step further in terms of stability, we studied precursor with linking between Cp ligand and alkylamide ligand. By comparing the two Cp-based heteroleptic precursors with the linking between Cp ligand and alkylamide ligand and the one without, we investigated the impact of linked ligand structure on growth characteristics, chemical compositions, crystallinity, and electrical properties. Furthermore, density functional theory (DFT) calculations were introduced for revealing the reaction energy and pathways.

AF-MoP-33 Role of Ligand Coordination Sphere on the Physico-Chemical Properties of New Mn Precursors: Structural, Thermal and DFT Investigations, Martin Wilken, Ruhr-Universität Bochum, Germany; A. Muriqi, M. Nolan, Tyndall National Institute, University College Cork, Ireland; A. Devi, Ruhr Universität Bochum, Germany

The interesting redox chemistry of manganese (Mn), its high abundancy in Earth's crust and additionally the associated unique electronic states make manganese-based materials like its oxides, nitrides or the sulfides suitable for a broad range of applications. In particular, when applied as thin films, Mn-based materials can function as cathodes or components in batteries^[1], electrocatalysts for oxygen evolution reaction (OER)^[2], dilute magnetic semiconductors (DMS) in spintronics^[3] etc. Atomic layer deposition (ALD) is one of the preferred methods to deposit thin films with a high degree of control of film properties. The precursors employed for ALD play an important role. Many of the known Mn precursors namely manganocene's (e.g., [MnCp₂])^[4], carbonyls [Mn₂(CO)₁₀]^[5], or β -diketones (e.g., [Mn(thd)₃])^[6] are associated with insufficient vapor pressure, poor shelf life and/or low thermal stability. One particular precursor which potentially overcomes the mentioned drawbacks is the all-nitrogen coordinated

manganese tert-butyl-acetamidinate [Mn(^tBuAMD)₂]^[7]. Motivated by this promising development, we further explored this class of Mn precursors by introducing the structurally related guanidinates and formamidinate ligand systems. This resulted in a series of new Mn complexes, which were thoroughly investigated by means of single-crystal XRD, nuclear magnetic resonance spectroscopy (NMR-Evans method), elemental analysis (EA), mass spectrometry (MS), and infrared spectroscopy (IR). To gain insights into the physicochemical properties, detailed thermal analysis was performed. Owing to the extreme sensitivity and reactivity of the compounds towards moisture and oxygen, density functional theory (DFT) calculations were performed to evaluate the origin of the pronounced reactivity. Based on the interesting results from the DFT studies, we investigated the influence of the manganese coordination sphere moving from all nitrogen to mixed O/N coordinated complexes, to optimize the physicochemical properties. The resulting Mn-ketoiminate class of compounds were successfully isolated and characterized in detail. A direct comparison of the all N- with the mixed O/N-coordinated Mn precursors in terms of the precursor characteristics relevant for ALD applications was performed which will be presented in this contribution.

AF-MoP-34 Surface Reactions During Nitrogen-Based Plasma Irradiation in Plasma-Enhanced Atomic Layer Deposition (PE-ALD) of Silicon Nitride, *Abdullah Y. Jaber*, Center for Atomic and Molecular Technologies - Osaka University, Japan; *M. Isobe, T. Ito, K. Karahashi, S. Hamaguchi,* Center for Atomic and Molecular Technologies, Osaka University, Japan

Plasma-enhanced atomic layer deposition (PE-ALD) plays an important role in uniform thin films deposition for semiconductor manufacturing with a large area and high accuracy. Silicon nitride (SiN) thin films are widely used in semiconductor devices. Typical technical challenges of developing PE-ALD of SiN (and any other materials) are to form the films with desired qualities at relatively low surface temperature, and possibly on specific surfaces exclusively (i.e., area selective ALD). In a typical SiN PE-ALD process, Si and chlorine (Cl) containing precursors such as dichlorosilane (DCS) H₂SiCl₂ are adsorbed on the surface in the first half-cycle, and in the second half-cycle, Cl atoms are removed and SiN is formed by nitrogen(N2) / hydrogen(H₂) or NH₃ plasmas. In the first half-cycle, the adsorption of precursors ends after a monolayer of Si is formed on the SiN surface with bonds terminated with Cl atoms, therefore the half-cycle is self-limiting. In the subsequent half-cycle, the plasma-based process is supposed to replace Cl on the surface with N and H atoms. The goal of this study is to understand the effects of nitrogen ions and hydrogen radicals on the removal of Cl atoms from the surface. We used molecular dynamics (MD) simulations to analyze how the CI-terminated surface interacts with incident N_{2⁺} ions and H radicals. The simulation results showed that most Cl atoms are removed with H radical irradiation with and without $N_{2^{+}}$ ions after H dose around $1.3x10^{16}$ (H atoms/cm²). Incident N_{2⁺} ions also remove Cl atoms from the surface while they nitride Si deposited on the surface, but the amount of CI removed from the surface is limited in this case. In addition, knock-on collisions transfer some CI atoms to a deeper layer, enhancing the Cl contamination of the surface. Incident NH3⁺ ions also remove more Cl atoms than N_2^+ ions only. This means that H plays an important role in converting desorbed precursors such as DCS to SiN in the N_2/H_2 or NH_3 plasma irradiation step. It was also confirmed that MD simulation results and available experimental observations showed good agreement.

AF-MoP-35 Effects of Oxidant and Substrate on Film Properties and Interlayer Formation of Atomic Layer Deposited ZrO₂ Films, *Seonyeong Park*, *S. Na*, Yonsei University, Korea; *Y. Lee*, Stanford University; *S. Chung*, *H. Kim*, Yonsei University, Korea

In the early stage of dynamic random access memory (DRAM) development, the thickness of the dielectric SiO₂ has been reduced in the Si based two dimensional structure to obtain high integration density and capacitance. However, as the DRAM devices have been continuously scaled down, thickness of SiO₂ reached a fatal limit of reaching the physical thickness at which leakage current due to tunneling effect occurs. To break through this, SiO₂ was replaced by high dielectric constant (k) materials[1]. Among the various high-k materials, ZrO₂ is one of the promising materials since it has good thermal stability, high dielectric constant (k~30) and wide bandgap (5.16-7.8 eV)[2,3]. However, when high-k thin film is deposited on TiN, which is widely used as an electrode, the electrical properties of the thin film are deteriorated such as increased leakage current density due to the interlayer formed between high-k film and TiN[4,5].

The interlayer formed between the substrate and dielectric film occurs due to the oxidation of the electrode during the film deposition process, which

is affected by the potential barrier height of the oxidation reaction[6]. Since the potential barrier height depends on the work function of the metal electrode, proper selection of the electrode is required. Studies have been reported on replacing the top electrode with Pt, Au, Ag, etc.[6], but these have the disadvantages of being expensive, which makes difficult to easily apply to real industry. Recently, Ru is recognized as a promising material because of its good thermal stability, low resistance, high work function, and relatively inexpensive among noble metals[7]. In this study, we used sputtered Ru as a top electrode and compared effects of bottom electrode using Ru and TiN on ZrO₂ film properties. In addition, to find the proper oxidant to reduce the oxidation of the substrate, the film properties using the two oxidants (oxygen plasma and hydroperoxide) were also compared and analyzed. For ultrathin high quality ZrO₂ film, atomic layer deposition (ALD) has been used, which can secure atomic layer controlled ZrO₂ film with high conformality and high uniformity.

*Corresponding author: hyungjun@yonsei.ac.kr

- [1] Chemical Reviews 99, 1823-1848 (1999)
- [2] Reports on Progress in Physics 69, 327-396 (2006)
- [3] Journal of Materials Science 53, 15237-15245 (2018)

[4] Journal of Vacuum Science & Technology B: Microelectronics and Nanometer Structure 27, 378 (2009)

[5] Solide-State Electronics 115, 133-139 (2016)

[6] Microelectronic Engineering 87, 98-103 (2010)

[7] Microelectronic Engineering 85, 39-44 (2008)

AF-MOP-36 Zinc Precursor Development for Future ALD Film Applications, Atsushi Sakurai, M. Hatase, M. Enzu, A. Yamashita, Y. Ooe, K. Takeda, R. Fukushima, ADEKA CORPORATION, Japan

ZnO has attractive physical properties such as being an intrinsically n-type semiconductor, having a wide bandgap and high electron mobility. Recently, we have seen an increasing number of stable p-type ZnO devices. Furthermore, 2TOC (two transistor zero capacitor) configuration using IGZO (indium gallium zinc oxide) transistor has been deemed a strong candidate for future 3D memory devices thanks to extremely low I_{off} of IGZO TFTs [1]. Diethyl zinc (DEZ) is the preeminent zinc precursor used to make many types of Zn-based ALD thin films such as zinc oxide, sulfide and nitride [2]. However, DEZ has poor thermal stability and can leave carbon impurities in zinc-based films grown at elevated temperatures. Furthermore, it has been pointed out that DEZ has an undesirable pyrophoric nature when exposed to air. Over the years, we have synthesized many kinds of zinc precursors (amide, alkyl, alkoxide, β -diketiminato, β -diketonato, etc.) which have a wide range of TGA delivery temperatures (Fig.1) and thermal stabilities in order to be compatible with future ALD applications. One interesting example of a new ALD Zn precursor is [Et₂NCH₂CH₂CH₂]₂Zn named as ZNP-2 whose structure has two zinc-carbon bonds with pendant amines at the ligand terminal. Fig.2 shows the cross-sectional SEM image of the ALD ZnO film which was made using the ZNP-2 precursor and H₂O as the co-reactant at 200°C. In addition to the ZNP-2 precursor, we will introduce more detailed structures and physical properties on the other zinc precursors as well as some ALD film data during the conference.

[1] Attilio Belmonte, IEDM2020, 28.2, [2] Shaibal Sarker, ALD2015

AF-MoP-37 Conformal Deposition of B_xC Thin Films from Triethylboron, Arun Haridas Choolakkal, H. Högberg, J. Birch, H. Pedersen, Linköping University, Sweden

Boron carbide (B_xC) films enriched in ¹⁰B is a promising neutron converter material for the next generation solid-state neutron detectors. Upon neutron irradiation, ¹⁰B produce detectable particles by the nuclear reactions ¹⁰B + n⁰ \rightarrow ⁷Li (0.84 MeV) + ⁴He (1.47 MeV) + Υ (0.48 MeV) and ¹⁰B + n⁰ \rightarrow ⁷Li (1.02 MeV) + ⁴He (1.78 MeV) that have 94% and 6% probability, respectively. Since the world has a ³He shortage, ¹⁰B solid-state detectors can potentially be a replacement to ³He detectors in large scale neutron facilities. For the next generation high resolution ¹⁰B detectors, films enriched in the ¹⁰B isotope must be deposited on pixelated sensor-chip substrates with high aspect-ratio morphologies. For such geometries, the currently employed magnetron sputtering technology is limited which highlight the need for alternative deposition (ALD). In addition, a low temperature process is required since the detector requires ohmic contacts which needs to be coated before converter layer deposition.

For coating high aspect-ratios, ALD would be the obvious choice, but the lack of ALD carbon precursors as well as carbide processes makes conformal continuous CVD processes a promising synthesis route. From our investigation, we report moderate temperature CVD of $B_{\boldsymbol{x}}C$ thin films on silicon substrates with 8:1 aspect-ratio morphologies, using triethylboron (TEB, ^{nat}B(C₂H₅)₃) as single source CVD precursor. Step coverage (SC) calculated from the cross-sectional scanning electron microscopy measurements shows that films deposited at ≤450 °C were perfectly conformal (SC = 1). We attribute this to the low reaction probability at low substrate temperatures enabling more gas phase diffusion into the features. The quantitative analysis using time of flight elastic recoil detection analysis (ToF-ERDA) and X-ray photoelectron spectroscopy (XPS) reveals that the as deposited films at 450 °C deposition temperature are B rich carbide material i.e., around 80% Boron, 15% carbon and 5% impurities, which is a promising B/C ratio for neutron detector application if ¹⁰B isotope enriched TEB is used. The density of the material, measured by X-ray reflectometry measurements, varies from 1.9 to 2.28 g/cm³ for the deposition temperatures varying from 425 to 550 °C in steps of 25 °C.

This study shows that $B_{*}C$ can be grown conformally by CVD at temperatures matching the thermal budget for the fabrication process and with a total B content which is on par with current state-of-the-art $^{10}B_{4}C$ films from neutron converters deposited by magnetron sputtering.

AF-MoP-38 Effect of Insulator-to-Metal Transition (Imt) in Vo₂ Deposited by ALD, Aline Jolivet, J. Cardin, C. Frilay, CIMAP Normandie Université, ENSICAEN, UNICAEN, CEA, CNRS, France; O. Debieu, CIRIMAT-INPT, France; P. Marie, S. Duprey, F. Lemarie, X. Portier, B. Horcholles, CIMAP Normandie Université, ENSICAEN, UNICAEN, CEA, CNRS, France; P. Bazin, LCS Normandie Université, ENSICAEN, France; J. More-Chevalier, Institute of Physics, Czech Academy of Sciences, Czechia; P. Fitl, Institute of Physics, Czech Academy of Sciences, Czechia; S. Cichoň, J. Lančok, Institute of Physics, Czech Academy of Sciences, Czechia; W. Jadwisienczak, School of Electrical Engineering and Computer Science, Ohio University; D. Ingram, Department of Physics and Astronomy, Ohio University; C. Labbé, CIMAP Normandie Université, ENSICAEN, UNICAEN, CEA, CNRS, France

Since its discovery in 1959, vanadium dioxide (VO₂) is well-known for its Insulator-to-Metal Transition (IMT) around 70 °C just above room temperature (RT). Over the last decade, it has gained interest because of its potential applications for resistive switching systems, energy storage in lithium-ion batteries, supercapacitors, infrared detectors, or thermochromic materials¹. Due to its low-temperature process, Atomic Layer Deposition (ALD) offers a real industrial advantage in the growth of VO₂, allowing to run deposition with low energy consumption and therefore low cost, and on substrates that can be temperature sensitive.

In this context, vanadium oxide films were deposited on both silicon (100) and glass substrates at 240 °C from vanadium tri-isopropoxide (VTIP) and water as an oxidizing agent. Films were grown with different thicknesses, ranging from 30 nm to 120 nm, and then annealed for one hour under forming gas (95% Ar, 5% H₂) at a temperature ranging from 400 °C to 550 °C by 50 °C steps.

According to a structural analysis held by X-ray diffraction (XRD) and transmission electron microscopy (TEM) as-deposited vanadium oxide films were amorphous and change morphology to polycrystalline with an admixture of VO₂ and V₂O₅ phases having crystallites size reaching 300 nm after annealing. The elemental analysis was performed by RBS and XPS studies will also be presented for a deeper understanding of their composition and stoichiometry.

VO₂ films were also characterized electrically and optically using resistivity measurements, and spectroscopic techniques such as Raman, FTIR, ellipsometry. All experimental results show a reversible and reproducible IMT around 70 °C for films annealed at 500 °C. For electrical measurements, the resistivity decreases down to 10⁻¹ Ω.cm upon IMT temperature. The optical ones show a particularly interesting result by FTIR spectroscopy on VO₂ films on silicon substrate, with an optical absorbance of 0.1 OD at RT which increases up to 0.9 OD above IMT temperature on a wavenumber range extending from 4000 to 1000 cm⁻¹. The phase transition is also observable in the UV-visible range on VO₂ films on glass substrate and is correlated with the appearance or disappearance of lowtemperature VO₂ peaks in Raman spectroscopy.

In conclusion, vanadium oxide films deposited by ALD were analyzed between RT and 100 $^\circ\text{C}$ and present promising properties tunable with

temperature, especially in the IR range, and paves the way for future applications in thermochromic materials.

1. Atomic layer deposition of vanadium oxides: process and application review. Mater. Today Chem. 12, 396–423 (2019).

AF-MoP-39 New ALD Processes for Y₂O₃ Using Molecularly Engineered Yttrium Formamidinates, *Sebastian Beer*, *N. Boysen*, Ruhr University Bochum, Germany; *A. Muriqi*, Tyndall National Institute, University College Cork, Ireland; *D. Zanders*, Ruhr University Bochum, Germany; *M. Nolan*, Tyndall National Institute, University College Cork, Ireland; *A. Devi*, Ruhr University Bochum, Germany

Yttrium oxide (Y_2O_3) thin films have been a subject of intensive research, particularly as an alternative high-k dielectric, due to its high relative permittivity (er ~17-20) and large bandgap of 5.5 eV. Furthermore, its high chemical resistivity and mechanical strength facilitate its application as a passivation layer in many fields.

Atomic layer deposition (ALD) has been established as one of the most promising techniques for the growth of high-quality layers for the above mentioned applications. As the precursor selection plays a pivotal role in an ALD process, the development of compounds with an optimal combination of volatility, reactivity and thermal stability is needed.

Besides the established Y precursors, such as β -diketonates or cyclopentadienyls, the all-N coordinated class of the amidinates¹ and guanidinates² has emerged as a promising class for the fabrication of yttrium-based materials. Recently, the structurally related formamidinate (famd) ligand class (*N*,*N*'-dialkyl-formamidinato) was explored, exhibiting a high volatility, reactivity, and stability.³

In this study, we focused on the systematic molecular engineering of Y formamidinates to fine-tune the physicochemical properties through a variation of the alkyl side chains. Among the four evaluated precursors, the *tert*butyl-substituted [Y(^tBu₂-famd)₃] showed an increased thermal stability and high reactivity towards H₂O, as revealed by thermal analysis and density functional theory (DFT) studies, respectively.

Subsequently, a thermal ALD process for Y_2O_3 using H_2O as co-reactant was developed, yielding dense *fcc*- Y_2O_3 films on Si substrates with smooth topography. Owing to the appealing structural, compositional and morphological quality of the layers, the process was used to deposit Y_2O_3 as a dielectric component in metal insulator semiconductor (MIS) capacitor structures.⁴ The promising electric properties set a strong platform for indepth studies to understand the interplay between precursor chemistry, ALD process development and integration in capacitor structures.

Literature:

¹ de Rouffignac, P., Park, J.-S., Gordon, R. G., *Chem. Mater.*, **2005**, 17, 19, 4808-4814.

² Mai, L., Boysen, N., Subasi, E., de los Arcos, T., Rogalla, D., Grundmeier, G., Bock, C., Lu, H.-L, Devi, A., *RSC Adv.*, **2018**, 8, 4987.

³ Boysen, N., Zanders. D., Berning, T., Beer, S. M. J., Rogalla D., Bock, C., Devi, A., *RSC Adv.*, **2021**, 11, 2565-2574.

⁴ Beer, S. M. J., Boysen, N., Muriqi, A., Zanders, D., Berning, T., Rogalla, D., Bock, C., Nolan, M., Devi, A., *Dalton Trans.*, **2021**, 50, 12944-12956.

AF-MOP-40 Nb/V-Doped Transparent Conductive TiO₂ Films Synthesized by ALD: Effect of Dopant Content and Growing Conditions, *Getaneh Diress Gesesse*, CIMAP ENSICAEN, France; O. Debieu, CIRIMAT, Université de Toulouse, CNRS, Université Toulouse, France; A. Jolivet, C. Frilay, S. Duprey, X. Portier, C. Dufour, P. Marie, C. Labbé, CIMAP Normandie Université, ENSICAEN, UNICAEN, CEA, CNRS, France; M. El-Roz, Laboratoire Catalyse et Spectrochimie, CNRS, ENSICAEN, Université de Caen, France; J. Cardin, CIMAP Normandie Université, ENSICAEN, UNICAEN, CEA, CNRS, France

Titanium dioxide (TiO_2) is a wide bandgap (3.0-3.20 eV) material that presents high transparency in the visible range; besides, it has a high refractive index. TiO_2 doping is widely investigated to induce structural and electronic modifications that may improve electronic properties and photocatalytic activities. Due to their additional electron compared to Ti and their atomic radius slightly above and below the one of Ti, Nb and V are the most promising candidates among the potential dopants of TiO_2 . In this context, this work aims to synthesize Nb/V-doped TiO_2 films by atomic layer deposition (ALD) with various dopant concentrations and growing conditions.

Nb/V-doped TiO_2 films were deposited on Si wafer and glass substrates with water as oxidizing agent using a shared ALD window for precursors niobium (V) ethoxide or vanadium (V) oxytriisopropoxide and titanium

isopropoxide, respectively. The dopant content was adjusted by the dopant ratio (R_{Nb} or R_V : range from 0 to 1), which is the number of dopant cycles over the total number of ALD cycles. Films were then annealed in N₂ or forming gas (95% N₂ and 5% H₂) at different temperatures of 400, 500, and 600 °C for 1 hour. To investigate the characteristics of films, various techniques, such as spectroscopic ellipsometry, spectrophotometry, Gl-XRD, Raman, FTIR, HRTEM, XRR, and four-probe resistivity were used for the as-deposited and annealed Nb/V-doped TiO₂ films.

It was observed that when introducing Nb or V into the TiO₂ matrix up to a doping ratio of 0.025, the thickness increased slightly compared to the undoped TiO₂ film, and then gradually decreased as the dopant ratio increases further. For both dopants, the refractive index and the electronic density of films were found to evolve similarly and in the opposite way to their thickness. In the as-deposited Nb-doped TiO₂ films with R_{Nb} = 0-0.025, the presence of the crystalline anatase phase was identified, but the peak intensity decreased and progressively shifted as R_{Nb} increased. In the case of the as-deposited V-doped TiO₂ film, only an amorphous phase was obtained which transform into crystalline phase with annealing.

After annealing, the optical transmittance and electrical resistivity of anatase phase crystallized films were measured for Nb/V-doped TiO₂ films. With increasing R_{Nb} , the films showed transmittance ranging from 60 to 80% in the visible range, which increase with the conductivity (~10² S.cm⁻¹). The optical and electrical properties of V-doped TiO₂ film are also performed and will be further discussed.

This work highlights the significant role of Nb and V dopants in tuning the structural, optical, and electrical property of TiO_2 .

AF-MOP-42 Atomic Layer Deposition of Tin Oxide Thin Films Using a New Liquid Precursor Bis(methylcyclopentadienyl) Tin, Makoto Mizui, N. Takahashi, F. Mizutani, Kojundo Chemical Laboratory Co., Ltd., Japan; M. Inoue, T. Nabatame, National Institute for Materials Science, Japan

Transparent conductive oxide thin films, including In-Sn-O (ITO), Zn-Sn-O (ZTO), and In-Zn-Sn-O (IZTO) films, have attracted attention for various applications such as flat-panel displays, sensors, and solar cells. We have reported the atomic layer deposition (ALD) of In_2O_3 and ZnO thin films using cyclopentadienyl-based precursors [1, 2]. In order to deposit such transparent conductive oxide thin films by ALD, ALD-Sn precursor is indispensable. This time, we report ALD of tin oxide (SnO_x) thin films using a new liquid cyclopentadienyl-based precursor.

Bis(methylcyclopentadienyl) tin, Sn(MeCp)₂, was synthesized as a liquid precursor. Differential scanning calorimetry (DSC) was conducted to measure its thermal decomposition temperature. The decomposition temperature was estimated approximately 220 °C, so the deposition temperature was set to 200 °C, which is the same temperature in the case of ALD of In₂O₃ and ZnO thin films [1, 2]. The vapor pressure of Sn(MeCp)₂ was determined by directly measuring equilibrium vapor pressures at several points. From the Clausius-Clapeyron plot for Sn(MeCp)₂, the precursor temperature was set to 130 °C, which corresponds to the vapor pressure of approximately 10 Torr.

SnO_x thin films were deposited on 150 mm Si wafers with native oxide films. ALD process was conducted by using Sn(MeCp)₂ as a precursor, and remote O₂ plasma as an oxidant. Linear growth of SnO_x thin film was observed when 0.1 s Sn(MeCp)₂ and 10 s O₂ plasma pulse times were applied. The growth per cycle (GPC) was approximately 0.06 nm/cycle with this ALD condition. This growth rate was rather slow compared with a previous experiment using tetrakis(dimethylamino)tin (TDMASn) as a precursor and remote O₂ plasma as an oxidant (~0.11 nm/cycle) [3]. One possible reason is that the deposition density was small due to steric hindrance of MeCp ligands. On the contrary, by applying H₂O for 10 s in the place of O₂ plasma, GPC was significantly small (~0.005 nm/cycle).

 SnO_x thin films were deposited by ALD using a new cyclopentadienyl-based precursor $Sn(MeCp)_2$, and linear growth of SnO_x thin film was confirmed.

References

[1] F. Mizutani, S. Higashi, M. Inoue, and T. Nabatame, AIP Advances 9, 045019 (2019).

[2] F. Mizutani, M. Mizui, N. Takahashi, M. Inoue, and T. Nabatame, ALD2021, AF1-10 (2021).

[3] M. A. Martínez-Puente, J. Tirado, F. Jaramillo, R. Garza-Hernández, P. Horley, L. G. S. Vidaurri, F. S. Aguirre-Tostado, and E. Martínez-Guerra, ACS Appl. Energy Mater. **4**, 10896 (2021).

AF-MoP-43 Growth Rate of Atomic Layer Deposition of Zinc Oxide Thin Films Using Bis(*n*-propyltetramethylcyclopentadienyl)zinc, Fumikazu Mizutani, M. Mizui, N. Takahashi, Kojundo Chemical Laboratory Co., Ltd., Japan; M. Inoue, T. Nabatame, National Institute for Materials Science, Japan

We have reported the atomic layer deposition of zinc oxide thin films using cyclopentadienyl-based precursors, bis(*n*-propyltetramethylcyclopentadienyl)zinc, Zn(Cp^{pm})₂, which is expected to deposit high-purity ZnO thin films [1, 2]. At that time, ZnO thin films were deposited at 200 °C on 150 mm Si wafers with native oxide films, and the ALD process consisted of alternating exposure to Zn(Cp^{pm})₂ and oxidants, H₂O followed by an O₂ plasma. However, the growth rate (GPC) was very small, about 0.02 nm/cycle or less. Until then, the precursor was supplied by vapor draw type or Ar assist type, so it was presumed that the reason for the small GPC was the low chemisorption coverage for low supply of the precursor. Therefore, we tried to increase the GPC by supplying the precursor by Ar bubbling type, which can supply a sufficient amount of precursor even at a low vapor pressure.

Other than supplying $Zn(Cp^{pm})_2$ by Ar bubbling, ZnO thin films were deposited on 150 mm Si wafers with native oxide films as before, and the ALD process consisted of alternating exposure to $Zn(Cp^{pm})_2$ and oxidants, H_2O followed by an O_2 plasma. As previously reported, the Clausius-Clapeyron equation for $Zn(Cp^{pm})_2$ is $log_{10}P$ (Pa) = -2502 / T (K) + 8.84, so the temperature of $Zn(Cp^{pm})_2$ was set to 70 °C where the vapor pressure of $Zn(Cp^{pm})_2$ is 0.3 Torr. The ZnO thin film was deposited for 30 cycles with the H_2O pulse time and the O_2 plasma pulse time fixed at 30 s and 15 s, respectively, and increasing the $Zn(Cp^{pm})_2$ pulse time from 5 s to 30 s. The film thickness of the obtained ZnO thin film increased as the pulse time increased, and was saturated around 20 s. The film thickness was about 7 nm when the Zn pulse time was 20 s to 30 s. Therefore, the GPC is about 0.2 nm/cycle or more, which is an extremely large increase compared to the previous GPC of about 0.02 nm/cycle or less.

Next, when a ZnO thin film was deposited with a $Zn(Cp^{pm})_2$ pulse time of 30 s, an H₂O pulse time of 30 s, and no O₂ plasma pulse, it was deposited without nucleation delay, and the film thickness was about 8 nm. Without the O₂ plasma pulse, the GPC would be slightly larger, as OH groups remained on the surface during the Zn(Cp^{pm})₂ pulse, increasing the amount of chemisorption.

Thus, if this $Zn(Cp^{pm})_2$ precursor is sufficiently supplied by Ar bubbling or the like, it can be deposited on a silicon wafer with a native oxide film without nucleation delay, and a large GPC of 0.2 nm / cycle or more can be obtained.

References

[1] F. Mizutani, S. Higashi, N. Takahashi, M. Inoue, and T. Nabatame, ALD2020, AF-MoP70 (2020).

[2] F. Mizutani, M. Mizui, N. Takahashi, M. Inoue, and T. Nabatame, ALD2021, AF1-10 (2021)

AF-MoP-44 Monte Carlo Simulation in a Laptop for Understanding Physical Interaction of Atomic Layer Deposition Precursors, *Bonwook Gu*, Incheon National University, Korea (Republic of); *N. Le, C. Nguyen*, Incheon National University, Viet Nam; *S. Yasmeen*, Incheon National University, Pakistan; *K. Youngho*, Incheon National University, Korea (Republic of); *H. Lee*, incheon National University, Korea (Republic of)

Since the reaction of atomic layer deposition (ALD) strongly depends on surface property, understanding of surface reaction mechanism between substrates and precursors is essential to predict and interpret thin film deposition in ALD. Recently, the many researchers have studied chemical reactions of ALD using density functional theory (DFT) and physical reaction using molecular dynamic (MD). However, DFT is suit for calculating a few of molecule adsorption but not simultaneous multiple adsorptions, and MD is not proper for a large scale simulation due to huge computing resource and long calculation time. In addition, although the steric hindrance effect between the molecules is an important physical factor for simulation of ALD, but it was not considered as a main variable for simulation. In this study, by adopting several assumptions and approximations, we developed a simple simulation method to understand physical steric hindrance effects of ALD precursors by using Monte Carlo (MC) without huge computing resources and applied the method to study surface reaction mechanism of ALD and area selective ALD (AS-ALD). We calculated the areal coverage of precursor on a specific surface used in ALD and AS-ALD using the MC simulation with a random adsorption model. The simulation results show high consistency agreement with experiment data. Based on the 2D model developed first, we extended the MC simulation to 3D system, and

obtained reliable results in bulky precursor systems. The simulation method developed in this study could be applied to many of ALD precursors and AS-ALD inhibitor systems just by using a laptop computer.

AF-MoP-45 Atomic Layer Deposition of Tin Monosulfide Films Using a Novel Cyclic Amide Divalent Metalorganic Precursor and H₂S, Mohd Zahid Ansari, School of Materials Science and Engineering, Yeungnam University, Korea (Republic of); *P. Janicek*, 2Institute of Applied Physics and Mathematics, Faculty of Chemical Technology, University of Pardubice, Czechia; *D. Nandi*, School of Materials Science and Engineering, Yeungnam University, Korea (Republic of); *B. Shong*, Department of Chemical Engineering, Hongik University, Korea (Republic of); *S. Kim*, School of Materials Science and Engineering, Korea (Republic of) of Materials Science and Engineering, Yourgnam University, Korea (Republic of); *S. Kim*, School of Materials Science and Engineering, Yourgnam University, Korea (Republic of) of Materials Science and Engineering, Yourgnam University, Korea (Republic of) of Materials Science and Engineering, Yourgnam University, Korea (Republic of) of Materials Science and Engineering, Yourgnam University, Korea (Republic of) of Materials Science and Engineering, Yourgnam University, Korea (Republic of) of Materials Science and Engineering, Yourgnam University, Korea (Republic of) of Materials Science and Engineering, Yourgnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Korea (Republic of) of Materials Science and Engineering, Yeungnam University, Ko

Tin (II) sulfide has emerged as an alternative light absorber and optoelectronics material, but its utilization is limited by lacking of purity due to various oxidation states of Sn. In this work, a better quality and stoichiometric SnS thin films were prepared by atomic layer deposition (ALD) using a novel metal-organic liquid precursor, N, N'-di (t-butyl)-2methyl-1,2-propanediamino tin(II) (Sn (dmpa)) and H₂S. The current ALD system showed distinctive ALD features, such as a self-limited growth, welldefined wide range ALD window between 75 to 175°C, and direct dependency on the number of ALD cycles on the film thickness. Interestingly, the proposed ALD-SnS process shows relatively high growth rate (GPC) value of 0.13 nm/cycle with no incubation delay at 125°C. DFT calculations are performed to investigate the surface reactions and selflimiting nature of the Sn precursor. The surface reactions are found to be feasible even at room temperature with either low- or high-coverage of the thiol group. The obtained films were characterized by several characterization techniques and found that the crystalline SnS films could be grown even at room temperature (25°C) to a high temperature of 250°C on a thermally grown SiO₂ substrate. Mix phase of cubic (π) and orthorhombic (o) SnS films are possible at below 100°C, whereas only orthorhombic phase is stable at above 125°C and SEM verified that the SnS films with different structures have different surface morphologies. The asgrown SnS film's purity was analyzed using Rutherford backscattering spectrometry, and X-ray photoelectron spectroscopy analyses, confirmed the deposition of a stoichiometric tin monosulfide (S/Sn=1) phase with negligible impurities. The optoelectronic properties (band-gap, refractive index, and extinction coefficient) of the SnS films were further evaluated via spectroscopic ellipsometry (SE) analysis. The results from the SE analysis supported the observed change from mixed $\pi\mbox{-}SnS$ and o-SnS to o-SnS with increasing temperature and allowed the determination of the energy bandgap (~1.1 eV) and a relatively broad semi-transparent window (up to 3000 nm). This newly ALD SnS process may have an immersive range of favorable prospects to utilize in temperatures compatible applications.

Acknowledgments: This work was supported by the National Research Foundation of Korea (NRF) grant funded by the Korean government (MSIT) (2021R1A2C1007601). The authors appreciated the financial support from grant LM2018103 by the Ministry of Education, Youth, and Sports of the Czech Republic. The precursor used in this study was provided by Hansol Chemical Co. Ltd., Korea.

AF-MoP-46 Density Functional Theory Study on the Selective Capping of Cobalt on Copper Interconnect, *Khabib Khumaini*, *R. Hidayat*, Sejong University, Korea (Republic of); *T. Mayangsari*, Universitas Pertamina, Indonesia; *T. Chowdhury*, *H. Kim*, Sejong University, Korea (Republic of); *S. Lee*, DNF Co. Ltd., Korea (Republic of); *W. Lee*, Sejong University, Korea (Republic of)

As the scaling of Cu interconnect systems continues, electromigration (EM) has emerged as a problem. The EM is mainly due to the movement of Cu atoms through the interface between Cu wire and the dielectric barrier. The selective Co capping on Cu improves the interface, resulting in increased EM lifetime. However, studies on the mechanism of the selective growth of Co on Cu over the inter-metal dielectrics are rare. The chemisorption of CpCo(CO)₂ on Cu and SiO₂ substrates was studied by density functional theory (DFT) calculation [1]. However, endothermicity on both substrates cannot explain the experimental reports [2]. Therefore, we studied the mechanism of selective Co growth using CpCo(CO)₂ on Cu over SiO₂ by DFT. We attempted to build reliable substrate models and consider all possible reaction pathways. We also investigated the roles of H₂ coreactant and the silvlation treatment of SiO₂. The calculation results show that H_2 plays a critical role on the Cu substrate but acts as a spectator on the SiO₂ substrates. The chemisorption was exothermic on Cu by forming H₂CO or CO, and H₂ lowered the activation energy from +1.09 eV to +0.62

eV. Contrarily, the reaction was endothermic on the -OH terminated SiO₂ releasing CO, which explains the origin of selective growth of Co on Cu. The chemisorption on the $-SiMe_3$ terminated SiO₂ surface showed the activation energy of +1.74 eV, which is higher than +1.35 eV on the -OH terminated SiO₂. Therefore, the silylation treatment can effectively improve the selectivity of CpCo(CO)₂ chemisorption on Cu over SiO₂, especially at high temperatures. Our DFT simulation successfully explains the mechanism of experimental observations on CVD Co.

[1] Adamczyk, A. J., et al. Computer Aided Chemical Engineering. 44. Elsevier, 2018. 157.

[2] Ryu, S. W., et al. Current Applied Physics. 16 (2016) 88.

AF-MoP-47 A New Low Temperature PEALD Process for HfO₂ Thin Films, Florian Preischel, D. Zanders, Ruhr University Bochum, Germany; A. Kostka, Ruhr-University Bochum, Germany; D. Rogalla, A. Devi, Ruhr University Bochum, Germany

With its high permittivity and large band gap of $E_g = 5.7 \text{ eV}$, HfO₂ is of significant interest for high- κ dielectric layers and excellent resistance ratios as well as fast switching speeds are reported for HfO₂ based memristor devices. To realize these microelectronic components, the deposition of pinhole-free thin films with an excellent uniformity and conformality is required. Owing to its saturative growth characteristics, atomic layer deposition (ALD) intrinsically fulfills these requirements. Plasma-enhanced ALD (PEALD) furthermore allows thin film depositions at low temperatures, as required for future flexible electronics, and is thus the method of choice. To enable the beneficial features of ALD, the physico-chemical properties of the precursor need to be carefully finetuned in order to optimize its thermal stability, volatility as well as reactivity. Looking for Hf precursors that fulfill these prerequisites, we investigated new heteroleptic Hf complexes. Starting from the parental Hf dialkylamide, we introduced a chelating formamidinate ligand that stabilizes and shields the Hf center, thereby increasing the thermal stability of the resulting complexe while retaining adequate reactivity and volatility. The resulting bis-(dialkylamido)-bis-(formamidinato) Hf(IV) precursor was synthesized on a multigram scale, structurally characterized and evaluated by thermogravimetric analysis and subsequently employed as a precursor for the deposition of HfO₂ in a PEALD process. Using an oxygen plasma, the typical ALD characteristics of precursor saturation, linearity and ALD temperature window were demonstrated with constant growth of 0.7 Å per cycle from 125 °C to 200 °C on Si(100) substrates. The resulting HfO2 films were further characterized by RBS/NRA, XPS and AFM, revealing the formation of pure and smooth HfO₂ layers. Compared to our previous work [1] with a closely related guanidinate precursor, shorter plasma pulses were sufficient to achieve ALD growth, preventing the formation of an interfacial SiO₂ layer, as revealed by transmission electron microscopy (TEM). By coating polyimide (PI) foils at temperatures as low as 150 °C, there is a potential of implementing the presented low-temperature HfO2 PEALD process into the development of flexible electronic devices in the future.

[1] D. Zanders et al., ACS Appl. Mater. Interfaces, 106 (2019) 28407

AF-MoP-48 Density Functional Theory Study on Atomic Layer Deposition of Silicon Nitride using Chlorosilane-type Precursors, *Tanzia Chowdhury*, *R. Hidayat, K. Khumaini, H. Son, H. Kim, W. Lee*, Sejong University, Korea (Republic of)

Silicon nitride thin film is an essential dielectric material in semiconductor devices. Recently, three-dimensional vertical NAND flash memory devices utilized it as a charge trapping and sacrificial layers. There have been a variety of silicon nitride ALD processes, including thermal and plasmaassisted schemes. Thermal ALD, which benefits from excellent step coverage, is typically achieved with chlorosilane-type precursors, such as dichlorosilane (SiH2Cl2) and hexachlorodisilane (Si2Cl6), using ammonia (NH₃) as the co-reactant. The reaction mechanism of thermal ALD was reported by experimental and computational studies [1,2]. However, most density functional theory (DFT) studies focused on a single reaction pathway for each precursor. For disilane or trisilane-type precursors having Si-Si bonds, the comparison between multiple reaction pathways is necessary, which is crucial to have a good insight into the film growth and resulting properties. Thus, in the present study, the comprehensive reaction mechanism of ALD silicon nitride using Si_2Cl_6 and Si_3Cl_8 was studied by DFT. The chemisorption of the silicon precursors was modeled and simulated on an NH₂/NH-terminated silicon nitride substrate. Multiple possible reaction pathways assuming the dissociation of a Si-Si or Si-Cl bond were considered, along with different initial orientations of the precursor molecule. For Si₂Cl₆, the most exothermic reaction pathway was

the cleavage of a Si-Cl bond to form $-NHSi_2Cl_5$ surface group along with a $-NH_3^*Cl^-$ as the byproduct. For Si_3Cl_8, the most exothermic pathway was the cleavage of a Si–Si bond, forming $-NHSiCl_3$ and Si_2Cl_5H byproduct. The activation energies of exothermic pathways will be calculated and compared to determine the most feasible pathway.

[1] L.L. Yusup et al., Appl. Surf. Sci., 432 (2018) 127.

[2]G. Hartmann et al, Phys. Chem. Chem. Phys., 20 (2018) 29152.

AF-MOP-49 PE-ALD of SiO₂ and Ti-doped SiO₂ in High Aspect Ratio Structures using Low Cost Precursors, Véronique Cremers, G. Rampelberg, A. Werbrouck, J. Dendooven, C. Detavernier, Ghent University, Belgium

PE-ALD of SiO₂ has been well studied in the past years, and has found a lot of applications as e.g. for advanced lithograpy, dielectric material in microelectronic devices, photovoltaics and optical applications. For these purposes where a high film quality is required in combination with a limited throughput, the use of more expensive precursors is justified. However, in recent years PE-ALD of SiO₂ found its way towards high surface applications as e.g. the encapsulation of OLEDS or deposition of anti-reflective coatings where precursor cost is a much bigger issue.

In this work, the PE-ALD growth characteristics of SiO₂ was investigated using four precursors in a different price setting: Bis(diethylamino)silane (BDEAS), (3-Aminopropyl)triethoxysilane (APTES),Tetraethyl orthosilicate (TEOS) and Hexamethyldisilazane (HMDS) (in order of high to low precursor cost price), in combination with oxygen plasma as a reactant.

Although it was possible to deposit SiO_2 with all four ALD processes, a significant difference in growth rate was observed. The growth rate varied from 1 Å/cycle for BDEAS to 0.2 Å/cycle for the TEOS-based process (Figure 1). Further the conformality of the four processes was investigated using macroscopic lateral structures with an equivalent aspect ratio (EAR) of 22. While BDEAS and APTES showed an excellent conformality, TEOS and HMDS showed a coated EAR of only 1 and 2.5 respectively. Here the coated EAR is defined as the EAR where the deposited film thickness equals 50% of the original film thickness. In order to enhance the growth rate and conformality of these last two processes, TiO_2 subcycles were added to the TEOS and HMDS- based processes using titanium(IV)isopropoxide (TTIP) as a low-cost precursor. Both Ti_xSi(1-x)O₂ films showed a strong improvement in growth rate, from 0.2 to 0.5 Å/cycle for the TEOS-based process and from 3.5 to 6.2 Å/cycle for the HMDS-based process, using a 1 TTIP :9 TEOS/HMDS subcycle ratio. Besides the growth rate, also a strong improvement in conformality was observed as is shown in Figure 2. The coated EAR for the TEOS-based process increased from 2.5 to 17.5 and for the HMDS based process from 1 to 12 (in both cases for a 1:9 (TiO₂:SiO₂ subcycle ratio)). These results could be promising when (no-pure) SiO₂ films are required for high surface area applications where a low precursor cost is relevant.

AF-MoP-50 Importance of XPS Investigations of ALD Grown 2D Materials, Jhonatan Rodriguez Pereira, R. Zazpe, J. Macak, University of Pardubice, Czechia

The success of graphene opened a door for an extensive research and utilization of semiconducting 2D transition metal dichalcogenides materials (TMDCs) displaying a range of unique properties [1]. Molybdenum disulphide (MoS2), a TMDCs benchmark material, has been widely studied for multiple applications. In parallel, 2D diselenide and ditelluride analogues, i.e. MoSe2 and MoTe2, have also attracted important interest due to their intriguing properties, such as a higher electrical conductivity than that of MoS2 among others [2, 3].

Recently, we have demonstrated the ALD synthesis of both 2D MoSe2 [4-7] and 2D MoTe2 [8] (using an in-house synthesized precursors), as well as their outstanding performances in different applications. XPS turned a key tool to provide detailed chemical composition analysis of as-deposited 2D Mo-based TMDCs on substrates of different nature. Besides, the post-performance XPS characterization was appealing since the applications of the aforementioned 2D materials involved chemical and/or electronic processes on the surface and it enabled to identify potential chemical composition changes and physico-chemical photo-electro stability of the 2D TMDCs.

This presentation will thus focus on the XPS as key tool for assessment of chemical composition of both as-deposited and post-performance 2D Mobased TMDCs, recent experimental results as well as the description of some inherent drawbacks that XPS must face during the analysis of the 2D materials. References:

[1] A. V. Kolobov, J. Tominaga, Two-Dimensional Transition-Metal,Dichalcogenides. Springer Series in Materials Science, SpringerInternationalPublishingAG, Switzerland2016

[2] D. Kong, H. Wang, J. J. Cha, M. Pasta, K. J. Koski, J. Yao, Y. Cui, Nano Lett. 2013, 13, 1341.

[3]	Α.	Eftekha	ari,	Appl.	Mat	er. 1	Today	201	7, 8.
[4]	R.	Zazpe	et	al,	FlatChe	em 20	020,	21,	100166
[5]	J.	Charvot	et	al,	Chempl	uschem	2020), 85	5, 576
[6]	J. Roc	lriguez-Per	eira	et al,	Surf. Sc	i. Specti	ra 202	0, 27,	024006
[7]	R. Za	zpe et al	, AC	S Appl	. Nano	Mater.	2021,	3, 12	, 12034
[8]	R. Z	azpe et	al,	Appl.	Mater.	Today	2021,	23,	101017.

AF-MoP-51 Atomic Layer Deposition for Modification of Various 1D Nanomaterials, Jan Macak, R. Zazpe, H. Sopha, University of Pardubice, Czechia; M. Rihova, Brno University of Technology, Czechia; L. Hromadko, S. Thalluri, University of Pardubice, Czechia

One-dimensional nanomaterials – materials with one dimension outside the nanoscale, further noted as 1D NMs – represent a class of very important nanomaterials with continuously increasing importance. Due to their intrinsic features, unique properties and diversity of functionalities, they count among the most widely studied materials nowadays. While considerable research efforts have been spent to synthesize various 1D NMs (e.g. nanopores, nanotubes or nanofibers), limited efforts have been devoted to surface modification and property tailoring of these materials.

However, it is their surface that comes into direct contact with various media (air, gases, liquids, solids) and influences the reactivity, stability and biocompatibility of these materials. The surface and aspect ratio (defined as their diameter to length ratio) influence the performance of these materials in various applications. Considering these facts, it is more relevant to tailor the surface of these materials and to be able to influence their properties and reactivity at the nanoscale, rather than to deal with tailoring their own bulk material composition.

The focus of this presentation is on the modification of two types of 1D nanomaterials – nanotubes and nanofibers. Numerous techniques can be utilized for this purpose, such as for example wet chemical or physical deposition techniques. However, it is only the Atomic Layer Deposition (ALD) that is capable of really uniform and homogenous coating of these 1D nanomaterials, in particular those of very high-aspect ratio.

The presentation will be mainly focused on modification of TiO₂ nanotube layers and various nanofibers of different aspect ratios via ALD.

Experimental details and some very recent application examples [1-10] and structural characterizations of these modified materials will be discussed.

1.	H. Sopha et al (2017), Appl. Mater. Today, 9, 104.
2.	H. Sopha et al. (2018), Electrochem. Commun.,97, 91.
3.	S. Ng et al. (2017), Adv. Mater. Interfaces, 1701146.
4.	F. Dvorak et al. (2019), Appl. Mater. Today, 14, 1.
5.	H. Sopha et al. (2019), <i>FlatChem</i> 17, 100130.
6.	M. Motola et al. (2019), Nanoscale 11, 23126.
7.	S. Ng et al. (2020), ACS Appl. Mater. Interfaces, 12, 33386.
8.	M. Motola et al. (2020) ACS Appl. Bio Mater. 3, 6447.
9.	M. Motola et al. (2021) Appl. Surf. Sci., 549, 149306.
10.	M. Rihova et al. (2021) Nanoscale Adv., 3, 4589.

AF-MoP-52 Atomic Layer Deposition of Cobalt(II) Fluoride Thin Films, *Elisa Atosuo*, *M. Mäntymäki*, *K. Mizohata*, *M. Chundak*, *M. Leskelä*, *M. Ritala*, University of Helsinki, Finland

Cobalt(II) fluoride, among some other first-row transition metal fluorides, is an excellent cathode candidate for lithium-ion and sodium-ion batteries. These metal fluorides have high theoretical potentials and high energy densities compared to the current oxide-based cathodes. However, although ALD is recognized in lithium-ion and sodium-ion battery research in general, the number of ALD processes for fluoride-based battery

materials has remained small. Particularly, these first-row transition metal fluorides have been lacking ALD processes. In this work, we present the first ALD process for cobalt(II) fluoride.

CoF₂ was deposited using CoCl₂TMEDA* and NH₄F as precursors. The films were characterized with XRD, EDS, XPS, ToF-ERDA, SEM, and AFM. The deposition temperature range studied was 180–300 °C, and all the deposition temperatures resulted in tetragonal CoF₂, as measured by XRD. Also XPS and ToF-ERDA confirm the films to consist of CoF₂. The impurity content measured with ToF-ERDA is low. Most importantly, the amounts of Cl and N, which are constituents of the precursors, are low, for example 0.53 and 0.08 at-% for a film deposited at 250 °C. The combination of a chloride-based precursor and NH₄F thus seems to work well in this case. The saturation of the growth per cycle with respect to pulse and purge lengths was confirmed at 250 °C, and the growth per cycle saturates to ~1.1 Å. In addition, the film thickness is linearly dependent on the number of applied cycles. Like many ALD metal fluorides, these films are rough, as seen in SEM and AFM. At a deposition temperature of 250 °C, for example, a ~60 nm film has a roughness of 12.6 nm.

*TMEDA: N,N,N',N'-Tetramethylethylenediamine

AF-MOP-53 Mechanistic Aspects of the Indium Nitride Growth under Atomic Layer Deposition Conditions: A Multiscale Modelling Study, Giane Damas, IFM, Sweden; K. Rönnby, Linköping University, IFM, Sweden; H. Pedersen, L. Ojamäe, Linköping University, Sweden

The wurtzite form from Indium Nitride has semiconducting behavior that, combined with advantageous electron transport properties, has offered potential applicability of this material in the field of electronics and lightemitting diodes.¹ The InN thin film is preferentially obtained using atomic layer deposition (ALD) techniques, with lower temperatures that are beneficial for the crystal stability and enable the utilization of ammonia precursor at such conditions.² In the process, trimethylindium is a wellknown In precursor that might undergo partial decomposition in the gas phase,³ resulting in the CH₃ radical release that is expected to affect the initial steps of the reaction mechanism. In addition, this precursor usually leads to high level of carbon impurities that is inconvenient for large scale production. Therefore, this project aims at fully understanding the mechanistic aspects of the adsorption and reaction-related processes leading to the In-rich layer formation for InN crystal growth by using a multiscale approach that comprises density functional theory (DFT) and Kinetic Monte Carlo (KMC) computational techniques.

The atomic-scale periodic calculations are carried out within the Perdew-Burke–Ernzerhof (PBE) level of theory in VASP⁴. Initially, the thermal effects are neglected to enable a more extensive investigation of the relevant reaction pathways. Such effects are then added to approximate the model to the actual experimental conditions (T= 593 K, 1 bar). The outcome indicates that the initial decomposition steps whether they occur in the gas phase or at the surface both lead to the final product, *i.e.* methylindium (MI) chemisorbed at the hcp site and ethane. However, the N- rich laver leads to an activation of this process that is found to facilitate the precursor dissociation at the surface environment, with an activation enthalpy $\Delta^{+}H$ <20 kJ/mol for TMI/DMI displacement towards other stable adsorption sites. In a second step, the hydrogen atoms are subsequentially removed through the involvement of two additional precursor molecules to produce low-weight hydrocarbons. The results also suggest the origin of the carbon impurities to be the CH₃ radical released during the process that in turn can form a strong chemical bond with the N-rich layer. From this point, all data necessary for the KMC simulation at the mesoscale level are acquired, which shall also be presented.

References:

1-Bhuiyan, A. et al. J. Applied Phys. 94, 2779 (2003).

2-Deminskyi, P. et al. J. Vac. Sci. Technol. A 37, 020926 (2019).

3-Hwang, J. et al. J. Electrochem. Soc., 155, 2 (2008).

4-Gresse, G. et al. Comput. Mat. Sci. 15, (1996).

AF-MOP-54 Simple Nanoscale Thermal Techniques for the Measurement of ALD Grown Thin Films, *Thomas Pfeifer*, S. Makarem, P. Hopkins, University of Virginia, USA

Measuring the thermal resistances associated with ALD / MBE grown films is critical for their design and use in the semiconductor industry, however traditional measurement techniques such as Time or Frequency Domain Thermoreflectance (TDTR and FDTR respectively) come with complications limiting their widespread adoption. In these traditional thermoreflectance techniques, two laser beams serve

to heat the sample (pump), and subsequently monitor changes in reflectivity (probe), which corresponds to changes in temperature. In TDTR, the beams are pulsed, and the temperature decay over time is measured. This requires the use of a precisely aligned and calibrated delay stage to temporally offset the pump and probe pulses, making this technique impractical for many. In FDTR, a pulsed or CW beam can be used, and the phase offset between sinusoidal heating and a sinusoidal temperature response is monitored. Difficulties in eliminating a frequencydependent systematic phase offset may make robust use of this technique difficult. in a more recently developed technique, Steady State Thermoreflectance (SSTR), the magnitude of the temperature response is monitored with varying pump intensity. This is limited to the measurement of one single parameter at a time however, and is typically only sensitive to the dominant thermal resistance in the system. SSTR also requires the use of a reference sample with an identical transducer in order to correlate a voltage as measured via the probe's photodetector to a known temperature rise.

We combine the principles behind SSTR (where the magnitude of the temperature response is roughly proportional to the net resistance), with that of FDTR (where the frequency-dependent response of the material system depends on all thermal properties). By examining the magnitude of the reflectivity response at multiple frequencies, we may now simultaneously measure multiple thermal parameters, such as thermal conductivity, volumetric heat capacity or thermal boundary conductance. This also enables the tightening of uncertainty of measured parameters. To validate this approach, we measure a variety of thin ALD-grown films (2-100nm) on several substrates (sapphire, silicon, diamond). All samples have an 80nm aluminum transducer deposited on top, and we measure the net thermal resistance across the ALD-grown film, in addition to the thermal conductivity of the substrate itself. We also measure bulk substrates, demonstrating the simultaneous measurement of thermal conductivity and heat capacity. We include robust uncertainty analysis to establish upper and lower bounds of measured values, and validate these results with an in-house TDTR system.

AF-MoP-55 Investigation of Tris(trialkylsilyl)Phosphides in Atomic Layer Deposition, Jaroslav Charvot, University of Pardubice, Czechia; M. Barr, J. Bachmann, University of Erlangen-Nürnberg, Germany; F. Bureš, University of Pardubice, Czechia

Phosphides are interesting materials among wide variety of scientific fields. InP and GaP are the most profound semiconductors with application in photovoltaics or electronics. Several ALD depositions of metal phosphides were already presented. Apart from one reported reaction of P(NMe₂)₃ with GaMe₃ affording GaP,^[1] PH₃ or its alkylated analogue *t*BuPH₂^[2]are usually used as a source of P^{-III} ion. Nevertheless, except high toxicity of PH₃, depositions using this precursor are often accompanied with lower reactivity supplemented by plasma activation^[3] or laser irradiation.^[4] PH₃ may also decompose during the deposition causing high content of phosphorus resulting in non-linear growth.

Trialkylsilyl ligand were utilized in ALD of As,^[5] Sb^[6] Se^[7,8] and Te^[7] several times. Its electropositive nature generates negative charge on the deposited atom ensuring high reactivity, while keeping good volatility and thermal stability. For example, tris(trimethylsilyl)phosphide is a favorite precursor for metal phosphide quantum dots. It is fairly volatile and can be distilled even at laboratory pressure. Interestingly, no ALD deposition using this class of precursor has been reported for so far. Therefore, this work aims to investigate tris(trialkylsilyl)phosphides as a potential ALD precursors. Preparation of these molecules will be discussed along with structure/thermal properties relationships and selected phosphides will be tested for thin film deposition in ALD.

[1] E. Graugnard, V. Chawla, D. Lorang, C. J. Summers, *Appl. Phys. Lett.* **2006**, *89*, 211102.

[2] N. Otsuka, J. Nishizawa, H. Kikuchi, Y. Oyama, J. Vac. Sci. Technol. A Vacuum, Surfaces, Film. 1999, 17, 3008.

[3] A. V. Uvarov, A. S. Gudovskikh, V. N. Nevedomskiy, A. I. Baranov, D. A. Kudryashov, I. A. Morozov, J. P. Kleider, *J. Phys. D Appl. Phys.*2020, *53*, 345105.

[4] M. Yoshimoto, A. Kajimoto, H. Matsunami, *Thin Solid Films*1993, 225, 70–73.

[5] T. Sarnet, T. Hatanpää, M. Laitinen, T. Sajavaara, K. Mizohata, M. Ritala, M. Leskelä, J. Mater. Chem. C2016, 4, 449.

[6] V. Pore, K. Knapas, T. Hatanpää, T. Sarnet, M. Kemell, M. Ritala, M. Leskelä, K. Mizohata, *Chem. Mater.***2011**, *23*, 247–254.

[7] T. Hantapää, V. Pore, M. Ritala, M. Leskelä, *Electrochem. Soc.*2009, *25*, 609–616.

[8] R. Zazpe, J. Charvot, R. Krumpolec, L. Hromádko, D. Pavliňák, F. Dvorak, P. Knotek, J. Michalicka, J. Přikryl, S. Ng, V. Jelínková, F. Bureš, J. M. Macak, *FlatChem*2020, *21*, 100166.

AF-MoP-56 Plasma Enhanced Atomic Layer Deposition of Nickel Oxide from Nickelocene: Growth Characteristics and Photoelectrochemical Performance, *Shane O'Donnell*, *F. Jose, M. Snelgrove*, *C. McFeely*, *R. O'Connor*, Dublin City University, Ireland

Owing to its relatively low cost and suitable band edge positions, n-type silicon has been widely investigated as a promising candidate for photoelectrochemical (PEC) water splitting photoanodes. However, its PEC performance is hampered due to rapid corrosion from the electrolyte solution rendering the silicon photoanode inoperable within a short period of use. NiO thin films deposited via atomic layer deposition (ALD) have been used as protective coatings to lessen this corrosion in an effort to enhance PEC performance through distancing the oxidation reaction from the n-Si photoanode surface and operating as an oxygen evolution reaction catalyst ¹.

We investigate growth kinetics of a plasma enhanced ALD process as well as the PEC performance of NiO thin films fabricated using nickelocene with O2 plasma co-reactant. We compare the performance of as-deposited NiO films with those reduced to Ni metal by including short H2 plasma steps during the deposition recipe, followed by a long H2 plasma anneal. Films are investigated in detail using quasi in-situ x-ray photoelectron spectroscopy (XPS), without vacuum break. PEC performance was measured to link the effects of varying film deposition parameters and processes on photocurrent output and film surface chemistry.

Through the investigation of various NiO film thicknesses and post deposition treatments, post deposition H₂ plasma annealing of all films thicknesses showed a reduction in photocurrent output and sample stability due to susceptibility to etching effects of the plasma treatment. From the work performed on supercycle deposition processes it is observed that the deposition of Ni films in continuous growth conditions with a post deposition H₂ plasma anneal results in films exhibiting greater contributions of Ni metal along with elevated carbon composition resulting from residual ligands being incorporated in the film during growth. Films where H2 exposure takes place during the growth process exhibit a lower relative carbon contribution due to the reduction of remnant ligands distributed throughout the film. The presence of Ni metal is also seen to be lower in sequentially grown films when compared to their post deposition annealed counterparts of the same number of ALD cycles

1 Ke Sun. etal. Energy Environ. Sci. 2012, 5 (7), 7872-7877

AF-MoP-57 Computational Investigations of Precursor and Deposition Chemistry in ALD and AS-ALD, *Ralf Tonner-Zech*, Leipzig University, Germany

The mechanistic details of many ALD processes are hardly understood. Despite the progress made by seminal work of Elliott and others, the chemical complexity of ALD processes requires the continued investigation of many aspects for the precursors and surface chemistry when the target is a comprehensive understanding by theoretical methods. We will show our recent progress made by applying density functional theory and electronic structure analysis approaches to understand aspects of ALD in more detail. Our aim is thereby always to cover the full periodicity of the growth surface to enable a realistic modelling of the experimental conditions. For selected aspects of the ALD process, gas phase screening approaches are suitable, which has been shown in the past and will be validated for our target systems here. We will also show that these investigations can be extended to area-selective ALD (AS-ALD) processes by small-molecule inhibitors (SMIs) and that theoretical modelling can play a crucial role in precursor and SMI design as well as uncovering the underlying principles.

AF-MoP-58 Epitaxial-like Growth of Ga₂O₃ Films on GaN Substrate by ALD using GaCp and Combinations of H₂O and O₂ Plasma Gas and Annealing Process, Toshihide Nabatame, M. Inoue, S. Soeya, T. Sawada, T. Onaya, A. Ohi, National Institute for Materials Science, Japan; M. Takahashi, K. Ito, Osaka University, Japan; N. Ikeda, National Institute for Materials Science, Japan; K. Tsukagoshi, National institute for Materials Science, Japan

 β -Ga₂O₃ is an attractive semiconductor because β -Ga₂O₃ has a large bandgap of 4.5-4.9 eV and a high breakdown electric field of 8 MV/cm for future power device. Large high-quality single-crystal β -Ga₂O₃ wafers can be generally fabricated by melt growth methods such as the floating zone and Czochralski techniques. Also, Ga₂O₃ films have been deposited on various substrates including Si and Al₂O₃ by sputtering, CVD and ALD methods. In ALD studies, Ga₂O₃ thin films were deposited by using various precursors such as Ga(acaC)₃, [(CH₃)₂GaNH₂]₃, Ga₂(NMe₂)₆, Me₂GaOiPr and Ga(CH₃)₃. In our previous study, we reported to deposit Ga₂O₃ thin films on GaCpl and combinations of H₂O and O₂ plasma as oxidant gas [1]. We also found that the Ga₂O₃ thin film had an amorphous structure and negligible small residual carbon. However, there has few report characteristics of ALD-Ga₂O₃ film on GaN.

In this paper, we investigated crystal growth of ${\rm Ga}_2 O_3$ films on GaN substrates by ALD and post-deposition annealing processes.

 Ga_2O_3 films were deposited on 3-inch (0001) c-plane GaN wafer (c-GaN) and (10-10) m-plane GaN with off-cut angles of 5° toward the (0001) (m-GaN) by ALD with GaCp and combinations of H₂O and O₂ plasma gas. ALD was performed at 300 °C and the growth per cycle value was 0.06 nm/cycle. 20-nm-thick Ga₂O₃ films on GaN substrates were annealed at 800 °C in N₂. The structure of as-deposited and annealed Ga₂O₃ films were evaluated using in-plane XRD and XRD pole figure measurement.

The as-deposited Ga₂O₃ film on c- and m- GaN substrates dominantly had an amorphous structure because of negligible small XRD peak. On the other hand, in c-GaN substrate, the annealed Ga₂O₃ film had four sharp peaks. These peaks at 2q = 19.0, 38.4, 59.3 and 82.4° were assigned to (-201), (-402), (-603) and (-804) of β-phase of Ga₂O₃, respectively. From the (-201) pole figure measurement, strong peak was also observed in the center. These indicated that epitaxial-like growth of β-Ga₂O₃ film with (-201) was dominantly formed on 3-inch GaN wafer. The epitaxial-like growth of β-Ga₂O₃ film on m-GaN was also observed.

This work was supported in part by JSPS KAKENHI (Nos. JP20H02189 and JP18J22998).

[1] F. Mizutani et al., J. Vac. Sci. Technol. A 38, 0022412 (2020).

AF-MoP-59 The Impact of Oxygen Reactants on ALD InOx films using novel dimethyl[N¹-(tert-butyl)-N²,N²-dimethylethane-1,2-diamine]indium precursor, *Seong-Hwan Ryu*, *T. Hong*, *S. Choi*, Hanyang University, Korea; *J. Seok*, *J. Park*, Hansol Chemical, Korea (Republic of); *J. Park*, Hanyang University, Korea

The oxide semiconductor has been attention due to its high mobility, low growth temperature, and low off-current characteristics (~10¹⁸ μ A/ μ m). In particular, there are several reports about the indium oxide (InOx) based materials such as IGZO, IZO, IGO, and IGTO, because it has low resistivity due to the large overlap between indium 5s orbitals forming the electron conduction path. Meanwhile, in terms of deposition technique of oxide semiconductor, atomic layer deposition (ALD) is attractive due to excellent control of the thickness and composition control during deposition. In this regard, several studies are reported depositing InOx with ALD, however, there are few studies about the growth mechanism such as precursor ligand and reactant. In this study, we deposited InOx using ALD with novel indium precursor and different reactants. The InOx grown with different reactants exhibits different material properties including crystallinity and electrical properties even at the same low growth temperature. For InOx analysis depending on different growth temperatures with the different reactants, we conducted X-ray photoelectron spectroscopy (XPS), X-ray diffraction (XRD), grazing-incidence wide-angle X-ray scattering (GI-WAXS), X-ray reflectometry (XRR) and Hall measurement.

AF-MoP-60 Development of an Innovative Method to Find New Efficient Gallium ALD Precursors, *P. Pavard*, CNRS-IPVF, France; *C. Gosmini*, LCM, École Polytechnique CNRS, France; *D. Lincot*, CNRS-IPVF, France; *V. Albin*, *V. Lair, A. Ringuede*, Chimie ParisTech, PSL University, CNRS, France; *A. Auffrant*, LCM, École Polytechnique CNRS, France; *Nathanaelle Schneider*, CNRS-IPVF, France

Atomic Layer Deposition (ALD) is a booming technology to deposit thin films and has been applied in several fields. This technique is based on

surface-chemical reactions, and relies on the gas phase transport of metal containing molecules into a reaction chamber. However, not any molecule is suitable to be used as precursor, as they must be thermally robust while being sufficiently volatile and chemically labile to react with the surface functional groups. Organometallic chemistry offers an infinite set of options to design new efficient precursors, though predicting their volatility and reactivity in the ALD chamber remains tricky.^[1] Establishing a new method to assess the physical and chemical properties of complexes would grant access to new ALD precursors and a better understanding of surface reactions.

This communication focuses on the development of new efficient gallium precursors to be used in the ALD of oxygen-free gallium-containing sulfide thin films.^[2] A series of gallium complexes with chelating nitrogen based ligands (guanidinate, amidinate and triazenides) were synthesized and characterized (NMR, XRD) thanks to modular procedures.^[3,4] To assess the thermal stability of the reagents and shed light on their transport in ALD, thermal analysis (TGA, DSC) were realized under N2 and vacuum to mimic transport conditions. Finally, reactivity studies in solution of established ALD precursors and synthesized complexes provide an insight of surface reactions which might take place in an ALD chamber.

References:

[1] S. E. Koponen, P. G. Gordon, S. T. Barry, Polyhedron2016, 108, 59-66.

[2] N. Schneider, M. Frégnaux, M. Bouttemy, F. Donsanti, A. Etcheberry, D. Lincot, Materials Today Chemistry2018, 10, 142-152.

[3] A. P. Kenney, G. P. A. Yap, D. S. Richeson, S. T. Barry, Inorganic Chemistry2005, 44, 2926-2933.

[4] S. Dagorne, R. F. Jordan, V. G. Young, Organometallics1999, 18, 4619-4623.

Characterization AF-MoP-61 Detailed of Bis(triisopropylcyclopentadienyl)strontium(Sr(iPr₃Cp)₂) for the Understanding of SrTiO3 Atomic Layer Deposition, Young Jae Park, Samsung Advanced Institute of Technology, Korea (Republic of); J. Park, Korea Research Institute of Standards and Science, Korea (Republic of); J. Han, J. Lim, B. Choi, Samsung Advanced Institute of Technology, Korea (Republic of); S. Kang, Korea Research Institute of Standards and Science, Korea (Republic of); C. Baik, Samsung Advanced Institute of Technology, Korea (Republic of) detailed characterization The of Bis(triisopropylcyclopentadienyl)strontium(Sr(iPr₃Cp)₂) precursor was conducted to understand growth mechanism in atomic layer deposition of SrTiO₃ thin films. First, the adsorption behavior was studied using an in-situ attenuated total reflectance-Fourier transform infrared spectroscopy (ATR-FTIR). The band of the Sr(iPr₃Cp)₂ spectrum on the Ge crystal surface was identical to that of the spectrum measured in the gas phase, but peak intensity was different. In addition, the absorption characteristics studies were carried out over the Ge crystal temperature in the range of 40~100°C. Upon increasing the temperature, a reduction of absorption was observed. Second, to investigate the volatility of Sr(iPr₃Cp)₂, vapor pressure curve was determined using thermogravymetric analysis. This method can reduce both precursor amount and time required for the vapor pressure measurement. Furthermore, molecular simulation was applied to explain the interrelationship between those properties evaluated in this study and molecular structure. Our study to understand the detailed behavior of precursor can be provided as useful information for optimization of ALD process and new precursor design.

AF-MoP-62 New Potential Synthesis Route of Molybdenum Nitride Nanosheets by Nitriding Molybdenum Disulphide Deposited by Atomic Layer Deposition (ALD), Julien Patouillard, SIMAP, Grenoble-INP, CNRS, France

The first syntheses of transition metal nitrides are derived from metallurgical processes and consisted in nitriding a powder of the metal or one of its oxides (Oyama, 1992; Toth, 1971). These nitrides were synthesized under severe conditions (T > 1200 °Cz and had low specific surfaces (Marchand et al., 1996). Subsequently, the development of catalytic applications requiring nitride powders with large specific surfaces made it necessary to use processes with more moderate temperatures (700 °C to 900 °C) (Marchand et al., 1996, 1999). These "softer" synthetic routes have developed and been applied to the formation of nitrides from transition metal sulfides.

Among the transition metal disulfides, molybdenum disulfide (MoS₂)is one of the most widely studied materials in recent years to synthesis molybdenum nitride due to its availability (E. R. Braitwaite & J. Haber, 1994). MoS₂ has a natural two-dimensional structure with the sandwichlike S-Mo-S layers serving as building blocks, in which the atoms in the layer are bonded with strong covalent bonding, while the layers are packed together with weak interlayer forces (Jariwala et al., 2014; Li & Zhu, 2015). In recent years, the emergence of 2D materials and the increase in the demand of metal nitrides nanosheets due to their remarkable physical and chemical properties such as high electrical conductivities, catalytic properties, energy storage, and conversion efficiency aroused a particular interest (Khazaei et al., 2013; Wang & Ding, 2018; Xiao et al., 2017, 2018; Zhong et al., 2016). Thereby, some research group(Sun et al., 2018) (Cao et al., 2020) have demonstrated the transformation of MoS₂ nanosheets exfoliated from bulk material into molybdenum nitride nanosheets with ammonia and urea reactive heat treatments, respectively.

In this work, we proposed a method to transform a well-controlled uniform MoS₂ thin film deposited by Atomic Layer Deposition into molybdenum nitride (MoN_x) nanosheets via an ammonia reactive heat treatment at 700 °C supported by in-situ reflectance measurements and ex-situ Raman and X-Ray Photoelectron spectroscopy characterizations.

AF-MoP-63 Phase-Transformation of as-Grown Crystalline VOx Films Using Ar-Plasma Annealing During Low-Temperature Hollow-Cathode Plasma-Assisted ALD Monitored via in-Situ Ellipsometry, Adnan Mohammad, S. Ilhom, University of Connecticut; A. Okyay, Stanford University; B. Willis, N. Biyikli, University of Connecticut

Vanadium oxide films show phase-change properties at different stoichiometries including the famous metal-insulator transition (MIT) for VO2around 70 °C shifting between monoclinic to tetragonal rutile structure phase with temperature change. Such layers have the potential to be used forlow-power electrical switches. The existing VOx ALD reports demonstrate mainly as-grown amorphous VOx films via thermal atomic layer deposition (ALD) using TEMAV and VTIP metal precursors and water vapor or ozone as co-reactant. These amorphous films are shown to transform in various crystalline phases using post-deposition annealing process at high temperatures (typically higher than 500 °C). However, no significant report is yet found on low-temperature as-grown VO2 films grown by thermal or plasma-ALD.

In this study we had two major goals: (i) To demonstrate as-grown crystalline VOx films using our customized hollow-cathode plasma-ALD reactor at substrate temperatures lower than 200 °C; (ii) to further improve the crystalline quality and transform the phase structure of the deposited VOx film into the desired VO2 stoichiometry. We have grown crystalline VOx thin films at substrate temperatures as low as 200 °C using TEMAV as the vanadium precursor and O₂ plasma as the oxygen co-reactant. The resulting as-grown film was crystalline V2O5. The recipe for the plasma-ALD experiments was as the following: 0.1 s of TEMAV pulse with 10 sccm of N2carrier flow, 50 sccm Ar-purge for 10 sec, 50 sccm O2 plasma for 10 s, plasma power 50- 300 W, followed by another 10 s of Ar purge. To provide enough TEMAV dose into the reactor chamber, the TEMAV precursor cylinder is heated at 110 °C. The resulting films are crystalline V_2O_5 with a growth per cycle (GPC) value reaching ~2 Å and a refractive index of 2.45. The corresponding growth process has been real-time monitored with insitu ellipsometry depicting the individual chemisorption and ligand exchange surface reactions.

Having accomplished our first goal, our next step is to incorporate in situ Ar-plasma annealing process steps to further improve and transform the crystal structure of $V_2 O_5$ films. Ar-plasma annealing parameters including Ar-flow rate, plasma annealing duration, and subsequent purge time will be investigated via in-situ process monitoring and ex-situ materials characterization. If successful, the resulting VO2 films will be characterized for their temperature-dependent electrical properties to validate the MIT behavior, paving the way for prototype switch devices.

AF-MoP-64 ALD Process Monitoring and Optimisation by Self-Plasma OES, Mantas Drazdys, Centrer for Physical Sciences and Technology, Lithuania; D. Astrauskyė, Center for Physical Sciences and Technology, Lithuania; R. Drazdys, Center fro Physcial Sciences and Technology, Lithuania; M. Audronis, Nova Fabrica Ltd., Lithuania

This work reports on the application of Self Plasma Optical Emission Spectroscopy (SPOES) to perform Process Gas Analysis (PGA) to monitor and optimize thin film atomic layer deposition (ALD) process, carried out using metalorganic precursors and water vapor or oxygen plasma as oxidizers. Depositions were carried out at 150°C using $N_{\rm 2}$ or Ar as carrier and purging gas. The ALD cycle consisted of four steps: (a) metalorganic precursor pulse, (b) purge, (c) oxidant pulse, and (d) purge. Purge times varied in the range of 2 - 120 seconds to find the optimal value based on the PGA results. We performed SPOES PGA using Broadband Plasma

Emission Monitoring (2B-PEM) of an inverted magnetron-based plasma reactor attached to the ALD process exhaust. The miniature plasma reactor can operate at pressures 7.5e–7 Torr to 7.5 Torr. The sensor signals derived from SPOES data react instantly to composition changes revealing trace amounts of constituents of the process material. We demonstrate how real-time process diagnostics, pump-down monitoring, process condition recognition and end-point detection, all taking place in parallel, facilitate ALD process yield maximisation and reaction by product residue in thin films prevention. Furthermore, the gas analysis technology demonstrated does not require additional (differential) pumping systems to perform analyses.

AF-MoP-65 Growth of Boron Nitride by Atomic Layer Deposition Using the in-Situ Decomposition of Ammonium Carbamate, Ana Álvarez-Yenes, M. Knez, CIC nanoGUNE, Spain

Boron nitride (BN) is a binary compound of boron and nitrogen alternatively linked which can exist in various crystalline forms, all of them analogous to the carbon allotropes. In each of its forms, BN presents interesting properties that make it a useful material in different applications, especially remarkable in nanotechnology.

The aim of this work is to develop and characterize a new Atomic Layer Deposition (ALD) process for the growth of boron nitride thin films. So far, this has been achieved by using ammonia as the nitrogen providing precursor in combination with a boron halide at temperatures above 400°C. The interest is to obtain boron nitride thin films at reaction temperatures below 275°C. In this work, ammonium carbamate (NH₄[H₂NCO₂]) and boron tribromide (BBr₃) are used as precursors. NH₄[H₂NCO₂] is a solid at room temperature which easily decomposes giving a CO₂/NH₃ mixture that can substitute ammonia, making the laboratory work safer and simpler. This work focuses on the analysis of the dependence of the growth rate of the process on its different parameters, aiming to optimize the deposition and predict the thickness of the grown films.

The thin films deposited using this process have been analyzed by means of X-Ray reflectivity, X-Ray photoelectron spectroscopy and electron energy loss spectroscopy among other techniques to conclude that the material deposited is amorphous boron nitride with a 1:1 stoichiometry.

AF-MoP-66 Modelling Atomic Layer Deposition of Noble Metals, Sylwia Klejna, AGH University of Science and Technology, Academic Centre for Materials and Nanotechnolog, Poland

The major challenge in metal ALD is the reduction process to yield the metallic target film from metal source that usually comprise of metal cation surrounded by anionic ligands. Existing strategies involve using reducing agent or, counterintuitively, oxidizing agent as second co-reagent. Using reducer as co-reagent, *e.g.* H₂, can lead to an abbreviated cycle, and reduced rate of deposition, when stable metal hydrides are not available. In process using oxidizing agent, *e.g.* O₂, transient metal oxide surface may be generated and that can greatly facilitate noble metal ALD [1]. In this scenario the reduction of metallic center is a result of precursor decomposition at this catalytic surface. The higher growth rate is thus achieved, because the metallic film forms also in the processes of combustion of ligands by the oxidizing agent, however with the danger of surface poisoning and oxide deposition.

Thus, the second co-reactant role is crucial. It is used to eliminate the surface bound species of the metal pulse and, at saturation, to produce a reactive overlayer – the catalytic oxide surface, which is characteristic for a particular noble metal. We therefore first investigate the thermodynamics to understand the self-limiting surface chemistry of the oxidizing co-reagent. We use density functional theory (DFT) to establish order of reactivity as a function of temperature and pressure of noble metals (Ru, Rh, Pd, Ag, Os, Ir, Pt, Au) to form oxides. Next, we examine the thermodynamics of ALD process that includes the transient generation of noble metal oxide.

Finally, we investigate reaction steps involved in the metal nucleation on the example of Pt ALD from MeCpPtMe₃ and O₂. We evaluate whether the nuclei of the catalytic surface can be formed during the O₂ co-reactant pulse, i.e. when oxidizing agent is introduced into the ALD chamber to combust hydrocarbon-based ligands into the volatile by-products (*e.g.* CO₂, H₂O). We discuss the possibility of production of transient surface bound OH groups predicted in previous study [2] and other by-products, *e.g.* CH₄ identified in the experiment [3]. The factors that facilitate nucleation are examined. This will allow to propose appropriate reagents and chemical processes to control and improve efficiency of the atomic layer deposition of noble metals.

- 1. The Journal of Chemical Physics, 2017, 146, 052822.
- 2. Langmuir, 2010, **26**, 9179-9182.
- 3. Physical Chemistry Chemical Physics, 2018, 20, 25343-25356.

AF-MoP-67 Effect of the N-Source on the Growth and Quality of Pe-ALD Tin Thin Films, *Clémence Badie*, Aix-Marseille University, France; *M. Barr, J. Bachmann,* Friedrich-Alexander-University Erlangen-Nürnberg (FAU), Germany; *T. Defforge, G. Gautier,* GREMAN, CNRS, Univ. Tours, France; *L. Santinacci,* Aix-Marseille University, France

The thermal ALD of TiN is a well-documented process. The common precursors combination is TDMAT/NH₃, for thermal as well as for plasmaenhanced ALD (PE-ALD). However, PE-ALD offers a larger variety of coreactant (N-sources): N2, H2 and N2/H2. Building the ALD window of TiN using the recipes from our manufacturer, the PE-ALD processes exhibit a longer cycle duration associated to a slower GPC regarding to the thermal ALD one. This is unexpected since the plasma should enhance the production of reactive species and then promote the deposition process. The GPC should therefore be higher. The aim of this work is to optimize the PE-ALD recipes using two different N-sources: NH_3 and N_2 . In addition to shorten the deposition duration, the effects of those N-sources as well as their dilution in Ar and the plasma power on the final properties of the films are also studied. The influence of those parameters has been monitored by in situ characterizations (ellipsometry and optical emission spectroscopy, SE and OES, respectively) and by ex situ characterizations (morphology, composition, crystalline structure and electric properties).

The recipe parameters are adjusted to limit the recombinations of the reactive species generated between the remote plasma source to the substrate. For instance, a large Ar dilution of both N2 and NH3 limits the film growth (lower GPC). A gas ratio of 1:1 for N-source and Ar flow is set to the optimal values. This is correlated with the OES spectra presenting that in diluted condition, the intensity of the Ar pics is predominant compared to the N-sources one. This suggests that high Ar dilution hinders the generation of N-reactive species leading to a reduction of the number of reactive species involved in the deposit growth and then, to lower GPC, mainly for N₂ plasma. The applied power (50 to 300 W) has no significant effect on the GPC with NH₃ plasma while, for N₂-based plasma, the GPC is maximum at the highest power. This is consistent with the expected low reactivity of N₂ (inert without plasma activation) as compared to highly reactive NH₃. Note that using NH₃, a thermal contribution cannot be discarded. Those results indicate that producing less active species facilitate their transport by limiting recombinations.

The films grown from both N-sources have a similar roughness, composition and morphology. However, the conductivity, conformality on high aspect ratio substrates (1:25) and the growth rate are better using NH_3 -based plasma. The N_2 plasma process exhibits an acceptable film quality and it should be considered as well since it uses a non-harmful gas.

AF-MoP-68 Silicon Nitride Grown by VHF (162 MHz) Plasma Enhanced Atomic Layer Deposition With Floating Multi-Tile Electrodes Using Di(Isopropylamino)Silane and N₂ Plasma, You Jin Ji, H. Kim, K. Kim, J. Kang, D. Kim, Sungkyunkwan University (SKKU), Korea (Republic of); K. Kim, Massachusetts Institute of Technology, Korea (Republic of); D. Kim, Sungkyunkwan University (SKKU), Korea (Republic of); A. Ellingboe, Dublin City University, Ireland; G. Yeom, Sungkyunkwan University (SKKU), Korea (Republic of)

To satisfy the requirements for applications of silicon nitride (SiN_x) at the highly scaled logic and memory devices, high quality films (i.e., low contamination, low roughness, etc.) with high conformality at low temperatures are demanded. To meet the stringent requirements for SiN_x applications, plasma enhanced atomic layer deposition (PEALD) is being extensively investigated as the deposition technique. However, the minimization of plasma damage and lowering the process temperature still remain as issues for SiN_x PEALD processes. In this work, the properties of PEALD SiN_xfilms deposited at a low process

temperatureof100°Cwithdi(isopropylamino)silane(DIPAS)andN2plasmaexcit edbyveryhighfrequency(VHF,162MHz)capactivelycoupledplasma (CCP) sources with a floatingmulti-tile type electrode and a conventional diode type electrode are investigated and compared in addition to the plasma characteristicsofbothplasmasources. ThePEALDSiNxfilmdepositedusing thefloating multi-tile

 $electrodeexhibited higher growth rate (~0.6 \text{\AA}/cycle), more stoichiometric film (N/Si~0.98),$

and higher conformality in a trench compared to those deposited by the convent i

Monday Afternoon, June 27, 2022

onal VHF

CCP. Inaddition, the improved electrical characteristics could be obtained with the efloating multi-tile electrode. The improved properties are believed to be related to the higher plasma density, higher radical density, and lower ion energy bombarding the substrate observed for the multi-electrode type through the enhanced power coupling efficiency between the pairs of multi-electrodes in the plasma source.

AF-MOP-69 Computational Fluid Dynamics Study of ALD Processes: Benchmarking and Validation, *Gizem Ersavas Isitman*, *D. Izbassarov*, *R. L. Puurunen*, *V. Vuorinen*, Aalto University, Finland

Gizem Ersavas Isitman 1 , Daulet Izbassarov 1 , Riikka L. Puurunen 2 , Ville Vuorinen 1

1. School of Engineering, Department of Mechanical Engineering, Aalto University

2. School of Chemical Engineering, Department of Chemical and Metallurgical Engineering, Aalto University

Atomic layer deposition (ALD) is an advanced gas-phase deposition technique [1]. ALD technology is exceptionally effective for obtaining highquality and conformal coating on high aspect ratio structures e.g in microelectronics, nano-materials, etc. These applications often require high manufacturing costs. Numerical methods provide a great opportunity to optimize the manufacturing process.

In this research, the purpose is to better understand ALD using computational fluid dynamics (CFD) in a porous medium by investigating how the reactive gas flow interacts with the catalytic surface. The ALD process is modeled using a CFD tool called aldFOAM [2]. The surface reactions are based on the irreversible Langmuir kinetics. The approach is validated against standard benchmark problems for non-reacting and reacting cases. First, the solver is validated against non-reacting flow past a single particle and tube banks (e.g Fig.1). Next, the irreversible Langmuir surface kinetics model in aldFoam solver is validated against reference data [3] for high aspect ratio structures. After the validation, the method is applied to investigate the surface coverage on different 2D and 3D shapes (e.g Fig.2) under a range of parameters such as pressure and Reynolds number (Re). It is found that the time to fully cover the surface decreases with increasing partial pressure and Re.

References

[1] J.R. van Ommen, A. Goulas, and R.L. Puurunen, "Atomic layer
deposition," in Kirk Othmer Encyclopedia of Chemical Technology, John
Wiley & Sons, Inc., 2021, 42 p.
https://doi.org/10.1002/0471238961.koe00059

[2] A. Yanguas-Gil, J.A. Libera, and J.W. Elam, Journal of Vacuum Science & Technology A: Vacuum, Surfaces, and Films 39, 062404 (2021).

[3] M. Ylilammi, O.M.E. Ylivaara, and R.L. Puurunen, Journal of AppliedPhysics123,205301(2018).

AF-MOP-70 Feature-Scale Conformality of Atomic Layer Deposition from Continuum to Free Molecular Flow: How Knudsen Number Influences Thickness Profile Characteristics, J. Velasco, Christine Gonsalves, G. Ersavas Isitman, J. Yim, E. Verkama, D. Izbassarov, V. Vuorinen, R. Puurunen, Aalto University, Finland

Atomic layer deposition (ALD) is often chosen over other techniques for thin film growth or surface modification because of its conformality, which originates from the self-terminating nature of the reactions used [1]. It is of paramount importance to understand how the conformality in high-aspectratio (HAR) surface features evolves with time and depends on process parameters and chemistry. Many simulation frameworks are available to model ALD growth in HAR features [2,4]: diffusion–reaction models, Monte Carlo models and ballistic models. Most simulation frameworks work in the free molecular flow conditions, where kinetic information of the reactions can be extracted from an experimental thickness profile by a simple slope method [3,4].

As seen in Figure 1, the thickness profile characteristics such as the halfthickness penetration depth $x_{50\%}$ and the slope at this half-thickness penetration depth, strongly depend on the Knudsen number in other diffusion conditions than free molecular flow. While $x_{50\%}$ can be taken as a simplistic measure for "conformality", the slope carries information of reaction kinetics, specifically of the sticking coefficient. To make interpretations on kinetics from experimental thickness profiles, understanding the flow conditions is of central importance. Specifically, assuming Knudsen flow when it is in reality not valid, would lead to incorrect (too high) interpretation of the sticking coefficient.

Recently [4], we showed that the way the process parameters affect the simulated thickness profile in HAR structures, depends on the diffusion regime: free molecular flow (Knudsen number Kn >> 1) has partly different trends than transition flow (Kn ~1). In this work, we extend the simulations to continuum conditions (Kn << 1). In addition to the previously used 1d diffusion–reaction model [4,5], in this work we also use computational fluid dynamics (CFD) calculations to investigate the processes in 3d.

This work was supported by the Academy of Finland (ALDI project).

References

[1] J.R. van Ommen, A. Goulas, and R.L. Puurunen, "Atomic layer deposition," in Kirk Othmer Encyclopedia of Chemical Technology, John Wiley & Sons, Inc., 42 p, (2021).https://doi.org/10.1002/0471238961.koe00059.

[2] V. Cremers et al., Applied Physics Reviews, 6(2), 021302, (2019).

[3] K. Arts et al., J. Vac. Sci. Technol. A, 37, 030908, (2019).

[4] J. Yim and E. Verkama et al., Phys. Chem. Chem. Phys., in press,

https://doi.org/10.1039/D1CP04758B

[5] M. Ylilammi, O. Ylivaara, and R.L. Puurunen, J. Appl. Phys., 123, 205301, (2018).

AF-MoP-71 Nucleation Studies of Copper ALD on SiO₂ and Si₃N₄, Krzysztof Mackosz, Empa - Swiss Federal Laboratories for Materials Science and Technology, Switzerland; A. Szkudlarek, AGH University of Science and Technology, Academic Centre for Materials and Nanotechnology, Poland; I. Utke, Empa - Swiss Federal Laboratories for Materials Science and Technology, Switzerland

Nucleation of ALD metal layers shows several key problems, among others, are nucleation delay and island growth, which prohibit the formation of thin compact metal films. To shed more light on the process of nucleation, we have performed a comprehensive electron microscopy study of the early growth stages of copper films prepared by means of the ALD protocol reported by the group of Karppinen [1] optimized for a hot-wall reactor.

Cu(dmap)₂ and HQ were used to grow copper by varying the growth temperature in the range of 120-160°C. Films were grown on Si substrates of different orientations covered with native oxide, on Si/SiO2 substrates with different thicknesses of oxide layer, and Si₃N₄ membranes. We have observed island growth irrespective of substrate type for a few hundreds of ALD cycles. Moreover, the growth mechanism is influenced by other factors such as the temperature of the substrate as well as the precursor flux. We will report on the coverage density, distribution of island size and chemical composition differs depending on growth conditions. We have resolved mixed compositions varying from metallic copper to copper oxide based on TEM studies. Both, Cu(I) and Cu(II) oxides are present in all the films studied. It is uncertain whether the oxide phase was deposited, or it is formed upon exposure to air during samples transfer from ALD reactor to TEM experimental chamber.

[1] D. J. Hagen, et al., Dalton Trans., 2018,47, 15791-1580

AF-MoP-72 Photocatalytic Degradation of Harmful Pollutants to Improve Indoor and Outdoor Air Quality, *Tobias Graumann, S. Pleger, C. Jacobs, C. Beyen, V. Sittinger,* Fraunhofer Institute for Surface Engineering and Thin Films IST, Germany

The application of photocatalytic materials in air treatment and air-borne pollutant remediation has been well established. Photocatalytically active materials used in urban areas on a commercial base can include paints, tiles, and concretes which mainly utilize TiO_2 as the photocatalyst material. Feasibility studies demonstrated the potential of those materials for the use of mineralizing organic compounds into carbon dioxide, water and corresponding mineral acids. Of special interest is the decomposition of nitrogen oxides with a main focus on NO and NO₂. A review of the photocatalytic effectiveness for outdoor applications is not always possible due to the simultaneous variation of other parameters, such as traffic density and weather conditions. Therefore, for the simulation of air pollutants numerical models for the release calculation in the atmosphere, the transport of pollutants in the gas phase, and the interaction with solid surfaces are used.

The determination of the deposition rate of available photocatalytic materials is currently limited to examining those embedded in matrices and formulations, e.g. for concrete surfaces, roof tiles or plaster, since these currently have the largest commercial proportion of photocatalytically active products. An evaluation of other, especially vacuum-based, coating processes for the deposition of TiO₂ layers has not yet taken place. To create a basic understanding of the essential process parameters influencing photocatalytic NO oxidation thermal atomic layer deposition is used.

In this work suitable process windows for the deposition of photocatalytic TiO₂ are identified and evaluated with a main focus on the precursor combinations TiCl₄/H₂O, TiCl₄/O₃, TTIP/H₂O, TTIP/O₃ and TiCl₄/TTIP. Process parameters affecting the crystallinity of the TiO₂ layers and thus the photocatalytic effectiveness and the process-related layer properties with different process temperatures and layer thicknesses are determined.

For a cross-method and unified comparison of the TiO_2 layer properties the photocatalytic oxidation of methylene blue in aqueous solution and the degradation of nitrogen monoxide in a photoreactor are compared with each other as well as the kinetic modelling which was observed and simulated during NO degradation.

AF-MoP-73 High Vacuum Chemical Vapour Deposition: High Growth Rate ALD-Like Film Synthesis and Epitaxial CVD for Integrated Photonics, *Wojciech Szmyt, J. Geler-Kremer,* Empa, Swiss Federal Laboratories for Materials Science and Technology, Thun, Switzerland; *S. Abel,* Lumiphase AG, Switzerland; *J. Fompeyrine,* Lumiphase, AG, Switzerland; *P. Hoffmann,* Empa, Swiss Federal Laboratories for Materials Science and Technology, Thun, Switzerland

In high-vacuum chemical vapour deposition (HV-CVD), heated substrates are exposed to continuous precursor fluxes from orifices in the precursor delivery system. Owing to the HV conditions, the precursor trajectories are ballistic, therefore the fluxes can be evaluated analytically. By blocking of individual effusion orifices, a range of different combinations of precursor fluxes can be explored in a single synthesis, which vastly accelerates the process optimization for the desired film properties. This approach is referred to as combinatorial deposition.

Moreover, due to HV conditions the probability of gas-phase collisions between precursor molecules is negligible, and thus, the chemical reactions occur strictly on the substrate surface. This characteristic allows for ALD-like synthesis of the films at a higher growth rate than ALD, provided that the substrate temperature is below the pyrolysis threshold of the precursors. In one of the previous works of our group (Reinke *et al.*, J. Phys. Chem. C, 2015, 119, 50, 27965–27971), we presented HV-CVD of TiO₂ using ALD precursors TTIP and H₂O at typical ALD temperatures 175-225°C, achieving growth rates of up to 2nm/min. The growth was demonstrated to follow ALD chemistry and kinetics. The combinatorial HV-CVD allowed to extract a range of kinetic parameters of the precursor system under study, thus proving the HV-CVD highly valuable as a tool for the fundamental study of ALD processes. Moreover, it shows that ALD processes can be adapted in the HV-CVD system for higher growth rates.

Owing to the negligible gas-phase reactions, in the CVD mode of HV-CVD, highly crystalline films are attainable. The HV-CVD-grown films rival the quality of molecular beam epitaxy results, achieved at lower temperatures, and providing much-improved process scalability and cost-efficiency. Our recent efforts have been focused on the epitaxial growth of BaTiO₃ on SrTiO₃-buffered substrates for integrated electro-optical devices using

Ba(iPr₃Cp)₂, TTIP and O₂ as precursors (Borzì *et al*, Materialia, 2020, 14, 100953). Employing the combinatorial mode of growth, we have established the optimal precursor fluxes for the correct stoichiometry, validated by Rutherford backscattering and Elastic recoil detection elemental analysis. We also demonstrated the films to be epitaxial by XRD as well as <0.5nm root mean square roughness, as shown by AFM, both of which being crucial for the application.

Author Index

— A — Aarik, L.: AF-MoP-14, 4 Aarnink, T.: AF-MoP-12, 3 Abel, S.: AF-MoP-73, 19 Ahmmad, B.: AF-MoP-11, 3 Ahn, J.: AF-MoP-19, 5; AF-MoP-20, 5 Albin, V.: AF-MoP-60, 15 Alderman, M.: AF-MoP-16, 4 Álvarez-Yenes, A.: AF-MoP-65, 17 An, J.: AF-MoP-5, 2 Ansari, M.: AF-MoP-45, 12 Astrauskyė, D.: AF-MoP-64, 16 Atosuo, E.: AF-MoP-52, 13 Audronis, M.: AF-MoP-64, 16 Auffrant, A.: AF-MoP-60, 15 — B -Bachmann, J.: AF-MoP-55, 14; AF-MoP-67, 17 Bačić, G.: AF-MoP-15, 4 Badie, C.: AF-MoP-67, 17 Bagrowski, D.: AF-MoP-13, 4 Bahrami, A.: AF-MoP-31, 8 Baik, C.: AF-MoP-61, 16 Barr, M.: AF-MoP-55, 14; AF-MoP-67, 17 Barry, S.: AF-MoP-15, 4; AF-MoP-16, 4 Bazin, P.: AF-MoP-38, 10 Becker, J.: AF-MoP-13, 4 Beer, S.: AF-MoP-39, 10 Beyen, C.: AF-MoP-72, 19 Birch, J.: AF-MoP-37, 9 Biyikli, N.: AF-MoP-17, 5; AF-MoP-63, 16 Blomberg, T.: AF-MoP-4, 1 Botan Neto, B.: AF-MoP-22, 6 Bouttemy, M.: AF-MoP-21, 5 Boysen, N.: AF-MoP-2, 1; AF-MoP-39, 10 Budzich, R.: AF-MoP-30, 8 Bureš, F.: AF-MoP-55, 14 — C — Cardin, J.: AF-MoP-38, 10; AF-MoP-40, 10 Chang, W.: AF-MoP-25, 6 Charvot, J.: AF-MoP-55, 14 Chaves, J.: AF-MoP-22, 6 Cheng, C.: AF-MoP-25, 6 Chiappim, W.: AF-MoP-22, 6 Choi, B.: AF-MoP-61, 16 Choi, S.: AF-MoP-59, 15 Choolakkal, A.: AF-MoP-37, 9 Chowdhury, T.: AF-MoP-46, 12; AF-MoP-48, 12 Chundak, M.: AF-MoP-52, 13 Chung, S.: AF-MoP-24, 6; AF-MoP-26, 7; AF-MoP-32, 8; AF-MoP-35, 9 Cichoň, S.: AF-MoP-38, 10 Ciuk, T.: AF-MoP-30, 8 Coutancier, D.: AF-MoP-21, 5 Cremers, V.: AF-MoP-49, 13 — D da Silva Sobrinho, A.: AF-MoP-22, 6 Dally, P.: AF-MoP-21, 5 Damas, G.: AF-MoP-53, 14 Debieu, O.: AF-MoP-38, 10; AF-MoP-40, 10 Defforge, T.: AF-MoP-67, 17 Dendooven, J.: AF-MoP-3, 1; AF-MoP-49, 13 Dentel, D.: AF-MoP-23, 6 Detavernier, C.: AF-MoP-3, 1; AF-MoP-49, 13 Devi, A.: AF-MoP-1, 1; AF-MoP-2, 1; AF-MoP-33, 8; AF-MoP-39, 10; AF-MoP-47, 12 Dill, P.: AF-MoP-23, 6 Dobrowolski, A.: AF-MoP-30, 8 Drazdys, M.: AF-MoP-64, 16 Drazdys, R.: AF-MoP-64, 16 Dufour, C.: AF-MoP-40, 10 Duprey, S.: AF-MoP-40, 10 Duprey, S.: AF-MoP-38, 10

Bold page numbers indicate presenter

— E -Ebert, S.: AF-MoP-23, 6 Eizenberg, M.: AF-MoP-6, 2 Ellingboe, A.: AF-MoP-68, 17 El-Roz, M.: AF-MoP-40, 10 Enzu, M.: AF-MoP-36, 9 Ersavas Isitman, G.: AF-MoP-69, 18; AF-MoP-70, 18 — F — Fitl, P.: AF-MoP-38, 10 Fompeyrine, J.: AF-MoP-73, 19 Franz, M.: AF-MoP-23, 6 Frégnaux, M.: AF-MoP-21, 5 Frilay, C.: AF-MoP-38, 10; AF-MoP-40, 10 Fukushima, R.: AF-MoP-36, 9 — G -Galvão, N.: AF-MoP-22, 6 Gautier, G.: AF-MoP-67, 17 Geler-Kremer, J.: AF-MoP-73, 19 Gesesse, G.: AF-MoP-40, 10 Gock, M.: AF-MoP-1, 1 Gonçalves de Lima, L.: AF-MoP-22, 6 Gonsalves, C.: AF-MoP-70, 18 Gosmini, C.: AF-MoP-60, 15 Graniel, O.: AF-MoP-2, 1 Graumann, T.: AF-MoP-72, 19 Gu, B.: AF-MoP-44, 11 — H — Hamaguchi, S.: AF-MoP-34, 9 Han, H.: AF-MoP-26, 7 Han, J.: AF-MoP-19, 5; AF-MoP-20, 5; AF-MoP-61, 16 Hatanpää, T.: AF-MoP-10, 3 Hatase, M.: AF-MoP-36, 9 He, S.: AF-MoP-31, 8 Hempel, W.: AF-MoP-13, 4 Hidayat, R.: AF-MoP-46, 12; AF-MoP-48, 12 Hintersatz, H.: AF-MoP-23, 6 Hirose, F.: AF-MoP-11, 3; AF-MoP-7, 2 Hoffmann, P.: AF-MoP-73, 19 Högberg, H.: AF-MoP-37, 9 Hong, T.: AF-MoP-59, 15 Hopkins, P.: AF-MoP-54, 14 Horcholles, B.: AF-MoP-38, 10 Hromadko, L.: AF-MoP-51, 13 -1-Ikeda, N.: AF-MoP-58, 15 Ilhom, S.: AF-MoP-17, 5; AF-MoP-63, 16 Ingram, D.: AF-MoP-38, 10 Inoue, M.: AF-MoP-42, 11; AF-MoP-43, 11; AF-MoP-58, 15 Isobe, M.: AF-MoP-34, 9 Ito, K.: AF-MoP-58, 15 Ito, T.: AF-MoP-34, 9 Izbassarov, D.: AF-MoP-69, 18; AF-MoP-70, 18 — J — Jaber, A.: AF-MoP-34, 9 Jacobs, C.: AF-MoP-72, 19 Jadwisienczak, W.: AF-MoP-38, 10 Jagiełło, J.: AF-MoP-30, 8 Janicek, P.: AF-MoP-45, 12 Jans, K.: AF-MoP-3, 1 Järvilehto, J.: AF-MoP-8, 2 Ji, Y.: AF-MoP-68, 17 Jin, Y.: AF-MoP-29, 7 Jõgiaas, T.: AF-MoP-14, 4; AF-MoP-28, 7 Jolivet, A.: AF-MoP-38, 10; AF-MoP-40, 10 Jose, F.: AF-MoP-56, 15 — к — Kang, J.: AF-MoP-68, 17 Kang, S.: AF-MoP-61, 16 Kang, W.: AF-MoP-19, 5; AF-MoP-20, 5 Kanomata, K.: AF-MoP-11, 3; AF-MoP-7, 2

Karahashi, K.: AF-MoP-34, 9 Kashyap, H.: AF-MoP-18, 5 Kei, C.: AF-MoP-25, 6 Khumaini, k.: AF-MoP-9, 3 Khumaini, K.: AF-MoP-46, 12; AF-MoP-48, 12 Kim, D.: AF-MoP-68, 17 Kim, H.: AF-MoP-24, 6; AF-MoP-26, 7; AF-MoP-32, 8; AF-MoP-35, 9; AF-MoP-46, 12; AF-MoP-48, 12; AF-MoP-68, 17; AF-MoP-9, Kim, K.: AF-MoP-68, 17 Kim, M.: AF-MoP-24, 6 Kim, S.: AF-MoP-45, 12 Kim, T.: AF-MoP-24, 6 Klejna, S.: AF-MoP-66, 17 Knez, M.: AF-MoP-65, 17 Korchnoy, V.: AF-MoP-6, 2 Kostka, A.: AF-MoP-47, 12 Kovalgin, A.: AF-MoP-12, 3 Krylov, I.: AF-MoP-6, 2 Kubota, S.: AF-MoP-11, 3 Kukli, K.: AF-MoP-28, 7 Kummel, A.: AF-MoP-18, 5 -L-L. Puurunen, R.: AF-MoP-69, 18 Labbé, C.: AF-MoP-40, 10 Labbé, C.: AF-MoP-38, 10 Lair, V.: AF-MoP-60, 15 Lančok, J.: AF-MoP-38, 10 Lawford, K.: AF-MoP-15, 4 Le, N.: AF-MoP-44, 11 Lee, H.: AF-MoP-44, 11 Lee, S.: AF-MoP-46, 12 Lee, T.: AF-MoP-25, 6 Lee, W.: AF-MoP-46, 12; AF-MoP-48, 12; AF-MoP-9, 3 Lee, Y.: AF-MoP-32, 8; AF-MoP-35, 9 Lemarie, F.: AF-MoP-38, 10 Leskelä, M.: AF-MoP-52, 13 Lim, J.: AF-MoP-61, 16 Lincot, D.: AF-MoP-60, 15 Liu, B.: AF-MoP-25, 6 — м — Mabuchi, T.: AF-MoP-29, 7 Macak, J.: AF-MoP-50, 13; AF-MoP-51, 13 Mackosz, K.: AF-MoP-71, 18 Magorian Friedlmeier, T.: AF-MoP-13, 4 Makarem, S.: AF-MoP-54, 14 Mändar, H.: AF-MoP-14, 4 Mäntymäki, M.: AF-MoP-52, 13 Marie, P.: AF-MoP-40, 10 Marie, P.: AF-MoP-38, 10 Masuda, J.: AF-MoP-15, 4 Mayangsari, T.: AF-MoP-46, 12 McFeely, C.: AF-MoP-56, 15 Miura, M.: AF-MoP-11, 3; AF-MoP-7, 2 Mizohata, K.: AF-MoP-52, 13 Mizui, M.: AF-MoP-42, 11; AF-MoP-43, 11 Mizutani, F.: AF-MoP-42, 11; AF-MoP-43, 11 Mohammad, A.: AF-MoP-17, 5; AF-MoP-63, 16 More-Chevalier, J.: AF-MoP-38, 10 Mpofu, P.: AF-MoP-27, 7 Muñoz-Rojas, D.: AF-MoP-2, 1 Muriqi, A.: AF-MoP-33, 8; AF-MoP-39, 10 — N — Na, S.: AF-MoP-26, 7; AF-MoP-32, 8; AF-MoP-35, 9 Nabatame, T.: AF-MoP-42, 11; AF-MoP-43, 11; AF-MoP-58, 15 Nagata, I.: AF-MoP-7, 2 Nandi, D.: AF-MoP-45, 12 Näsi, M.: AF-MoP-4, 1 Nguyen, C.: AF-MoP-44, 11

Author Index

Nielsch, K.: AF-MoP-31, 8 Nisula, M.: AF-MoP-3, 1 Nolan, M.: AF-MoP-33, 8; AF-MoP-39, 10 -0-Obenlüneschloss, J.: AF-MoP-1, 1 O'Connor, R.: AF-MoP-56, 15 O'Donnell, S.: AF-MoP-56, 15 Ohi, A.: AF-MoP-58, 15 Ojamäe, L.: AF-MoP-53, 14 Okyay, A.: AF-MoP-17, 5; AF-MoP-63, 16 Onaya, T.: AF-MoP-58, 15 Ooe, Y .: AF-MoP-36, 9 — P — Park, I.: AF-MoP-5, 2 Park, J.: AF-MoP-59, 15; AF-MoP-61, 16 Park, S.: AF-MoP-24, 6; AF-MoP-26, 7; AF-MoP-32, 8; AF-MoP-35, 9 Park, Y.: AF-MoP-61, 16 Patouillard, J.: AF-MoP-62, 16 Pavard, P.: AF-MoP-60, 15 Pedersen, H.: AF-MoP-37, 9; AF-MoP-53, 14 Petit, R.: AF-MoP-3, 1 Pfeifer, T.: AF-MoP-54, 14 Pietak, K.: AF-MoP-30, 8 Piirsoo, H.: AF-MoP-28, 7 Pleger, S.: AF-MoP-72, 19 Portier, X.: AF-MoP-40, 10 Portier, X.: AF-MoP-38, 10 Preischel, F.: AF-MoP-47, 12 Puurunen, R.: AF-MoP-70, 18; AF-MoP-8, 2 — R — Radu, I.: AF-MoP-25, 6 Rampelberg, G.: AF-MoP-49, 13 Ren, X.: AF-MoP-23, 6 Rihova, M.: AF-MoP-51, 13 Ringuede, A.: AF-MoP-60, 15 Ritala, M.: AF-MoP-10, 3; AF-MoP-52, 13 Ritslaid, P.: AF-MoP-28, 7 Ritter, D.: AF-MoP-6, 2 Rodriguez Pereira, J.: AF-MoP-50, 13 Rogalla, D.: AF-MoP-47, 12 Roh, H.: AF-MoP-9, 3

Author Index

Rönnby, K.: AF-MoP-53, 14 Ryu, S.: AF-MoP-59, 15 — S — Saito, K.: AF-MoP-11, 3; AF-MoP-7, 2 Sakurai, A.: AF-MoP-36, 9 Salari Mehr, M.: AF-MoP-14, 4 Santinacci, L.: AF-MoP-67, 17 Sávio Pessoa, R.: AF-MoP-22, 6 Sawada, T.: AF-MoP-58, 15 Schneider, N.: AF-MoP-21, 5; AF-MoP-60, 15 Seok, J.: AF-MoP-59, 15 Shiotani Marcondes, M.: AF-MoP-22, 6 Shong, B.: AF-MoP-45, 12 Shukla, D.: AF-MoP-17, 5 Sittinger, V.: AF-MoP-72, 19 Snelgrove, M.: AF-MoP-56, 15 Soeya, S.: AF-MoP-58, 15 Son, H.: AF-MoP-48, 12; AF-MoP-9, 3 Sopha, H.: AF-MoP-51, 13 Spiegelman, J.: AF-MoP-18, 5 Spiering, S.: AF-MoP-13, 4 Su, C.: AF-MoP-25, 6 Su, Y.: AF-MoP-25, 6 Szkudlarek, A.: AF-MoP-71, 18 Szmyt, W.: AF-MoP-73, 19 - T -Takahashi, M.: AF-MoP-58, 15 Takahashi, N.: AF-MoP-42, 11; AF-MoP-43, 11 Takeda, K.: AF-MoP-36, 9 Tamm, A.: AF-MoP-28, 7 Tarre, A.: AF-MoP-14, 4 Tegenkamp, C.: AF-MoP-23, 6 ter Veen, R.: AF-MoP-4, 1 Thalluri, S.: AF-MoP-51, 13 Tokumasu, T.: AF-MoP-29, 7 Tonner-Zech, R.: AF-MoP-57, 15 Tröger, J.: AF-MoP-4, 1 Tsukagoshi, K.: AF-MoP-58, 15 Uene, N.: AF-MoP-29, 7 Utke, I.: AF-MoP-71, 18

-vvan der Zouw, K.: AF-MoP-12, 3 van Duin, A.: AF-MoP-29, 7 Vandenbroucke, S.: AF-MoP-3, 1 Velasco, J.: AF-MoP-70, 18; AF-MoP-8, 2 Vereecken, P.: AF-MoP-3, 1 Verkama, E.: AF-MoP-70, 18; AF-MoP-8, 2 Vihervaara, A.: AF-MoP-10, 3 Vincent, T.: AF-MoP-21, 5 Vos, R.: AF-MoP-3, 1 Vuorinen, V.: AF-MoP-69, 18; AF-MoP-70, 18 — w — Wang, H.: AF-MoP-25, 6 Weber, M.: AF-MoP-2, 1 Weinfeld, K.: AF-MoP-6, 2 Werbrouck, A.: AF-MoP-49, 13 Wilken, M.: AF-MoP-33, 8 Willis, B.: AF-MoP-17, 5; AF-MoP-63, 16 Wohnhaas, T.: AF-MoP-13, 4 Wysmołek, A.: AF-MoP-30, 8 - X -Xu, X.: AF-MoP-6, 2 — Y — Yalon, E.: AF-MoP-6, 2 Yamashita, A.: AF-MoP-36, 9 Yang, I.: AF-MoP-5, 2 Yasmeen, S.: AF-MoP-44, 11 Yasuhara, S.: AF-MoP-29, 7 Yeom, G.: AF-MoP-68, 17 Yim, J.: AF-MoP-70, 18; AF-MoP-8, 2 Yoon, H.: AF-MoP-32, 8 Yoon, S.: AF-MoP-5, 2 Yoshida, K.: AF-MoP-11, 3; AF-MoP-7, 2 Youngho, K.: AF-MoP-44, 11 — z — Zaitsu, M.: AF-MoP-29, 7 Zanders, D.: AF-MoP-1, 1; AF-MoP-39, 10; AF-MoP-47, 12 Zazpe, R.: AF-MoP-50, 13; AF-MoP-51, 13