

In-situ and in-vacuo studies on area selective atomic layer deposited ruthenium films on silicon and silicon oxide

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We present a thermal activated inherent area selective atomic layer deposition (*thALD*) of ruthenium (*Ru*) on crystalline silicon (*cSi*) wafer and silicon oxide (*SiO₂*) interfaces.

The selective deposition is based upon the inhibited nucleation of ruthenium on oxide surfaces such as *SiO₂* compared to *Si* and metal surfaces. ThALD ruthenium (5-20 nm thick) deposited with the organometallic precursor ECPR [(ethylcyclopentadienyl)(pyrrolyl) ruthenium(II)] and molecular oxygen [1, 2] on 4 inch silicon [100] wafers with a pattern of *cSi* dipped by diluted hydrofluoric acid (*HF*), and native silicon oxide (1.8 nm).

In our experiments, the pattern on the wafer was created by lithography with AZ[®] 5214 E resist. A dip with 0.5% *HF* (30 s) was used to remove the native oxide film on *cSi* and to create a hydrogen-terminated surface. Immediately afterwards, the resist was stripped by a treatment with acetone, 2-propanol, and clean water (conductivity 0.05 $\mu\text{S}/\text{cm}$). After drying, the wafer was transferred into ALD tool for deposition within less than 5 minutes.

As shown in [2-4], the initially incubating and nucleation periods strongly depend on the deposition temperature. On *HF*-dipped *cSi* after a nucleation period of 10 ALD cycles a steady-state *Ru*-on-*Ru(Ox)* growth with a GPC of ca. 0.9 \AA was observed (Fig. 1) at 180 °C. In the linear steady-state region the GPC was nearly independent on the deposition temperature [2]. On native *SiO₂*, only isolated islands of *Ru* were formed in negligible quantity after 150-180 ALD cycles, consisting of non-stoichiometric *Ru* / *Ru(Ox)*. Here, growth started after 180 ALD cycles, whereas on *HF*-dipped *Si* a 16 nm thick *Ru* film ($R_s = 14,2 \Omega/\square$; $\rho = 22,72 \mu\Omega \cdot \text{cm}$) has been deposited already (Fig. 2). A higher quality of selectivity became be achieved by combining ALD with selective etching using an *O₂* or *O₃* purge after a certain number of ALD cycles (Fig. 3). As we demonstrated earlier [1, 2], purge steps with molecular hydrogen (*H₂*) during *Ru*-ALD can prevent blister formation.

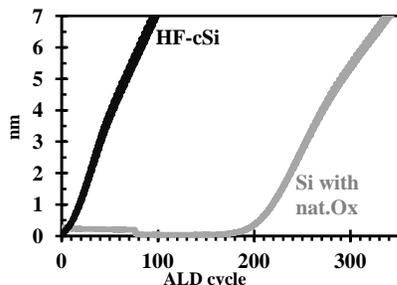


Fig. 1: In-situ real-time spectroscopic ellipsometry of selective ALD a) *HF* dipped *cSi* b) *Si* with nat. *SiO₂*; deposition temp. 180 °C

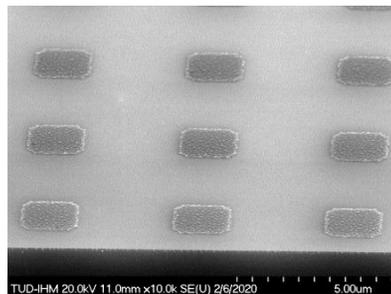


Fig. 2: SEM images of 16 nm *Ru* [180x *Ru* (ECPR-*O₂*) *thALD*; 180 °C] on *Si* wafer with a pattern of crystalline silicon dipped by *HF* and untreated native silicon dioxide (1.8 nm) - *Ru* grew on *cSi* only

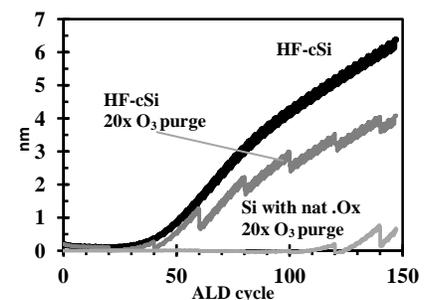


Fig. 3: In-situ real-time spectroscopic ellipsometry of selective ALD with an additional *O₃* Purge after 20 ALD Cycle; Substrates: a) *HF* dipped *cSi*; b) *HF* dipped *cSi* and c) *Si* with nat. *SiO₂*

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