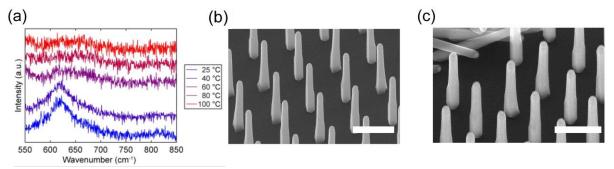
## Oxidation State Discrimination in the Atomic Layer Deposition of Vanadium Oxides

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Tunable control over the oxidation state of transition metal oxides (or sulfides) remains a significant challenge in ALD. Traditionally, the oxidation state of the metal precursor or strength of oxidant dictates the final oxidation state of the metal, and stoichiometry of elements, in the resulting film. In this work, a single vanadium  $3^+$  precursor performs self-limiting growth of thin films that span common oxidation states of vanadium,  $3^+$ ,  $4^+$ , and  $5^+$ . ALD of V<sub>2</sub>O<sub>3</sub>, VO<sub>2</sub>, and V<sub>2</sub>O<sub>5</sub> are realized via *four* distinct reaction mechanisms accessed at the same temperature by judicious choice of oxygen sources and co-reactants. Saturation behavior in all cases, the O<sub>3</sub>, H<sub>2</sub>O<sub>2</sub>, H<sub>2</sub>O/O<sub>2</sub>, and H<sub>2</sub>O<sub>2</sub>/H<sub>2</sub> processes, is observed by *in situ* quartz crystal microbalance (QCM) and surface chemistry is studied with quadrupole mass spectrometry. As expected O<sub>3</sub> oxidizes vanadium fully to the 5<sup>+</sup> state to grow crystalline V<sub>2</sub>O<sub>5</sub>. Growth with O<sub>2</sub> has not observed and self-termination with H<sub>2</sub>O occurs after less than 250 cycles corresponding to film growth of ~ 5 nm as measured by spectroscopic ellipsometry. When H<sub>2</sub>O and O<sub>2</sub> are introduced in sequence, synergistic growth of amorphous V<sub>2</sub>O<sub>5</sub> occurs. In this synergistic process, a clear distinction between non-oxidative protic ligand exchange, the H<sub>2</sub>O reaction, and metal oxidation, the O<sub>2</sub> reaction, is demonstrated in either reaction order. In addition, it is apparent from these results that this vanadium precursor requires an oxidation mechanism in sustained ALD film growth.

In an extension of this approach, we can produce films with the intermediate vanadium  $4^+$  oxidation state, VO<sub>2</sub>, through reducing equivalents – H<sub>2</sub> gas. In traditional A/B-type ALD, H<sub>2</sub>O<sub>2</sub> produces an amorphous mixed  $4^+/5^+$  oxide that is easily reduced to vanadium  $3^+$  oxide, V<sub>2</sub>O<sub>3</sub>, or oxidized to vanadium  $5^+$  oxide, V<sub>2</sub>O<sub>5</sub>, by varying the atmosphere of a low temperature anneal. Phase pure VO<sub>2</sub> is not readily accessible. When H<sub>2</sub> is dosed after H<sub>2</sub>O<sub>2</sub> during growth, amorphous films of VO<sub>2</sub> are grown that can be crystallized with a low temperature, 350 °C, anneal. These VO<sub>2</sub> films show the characteristic temperature dependent Raman spectroscopy response in the expected temperature range, Figure 1, of the well-known insulator to metal transition (IMT) for VO<sub>2</sub>. Conformal films of VO<sub>2</sub> were grown on indiumtin oxide (ITO) nanorods in this method, Figure 1, in order to modify the phonon response of the ITO nanorods. A distinct blue-shift of the wavelength of all absorption spectral features above the IMT transition temperature is observed and can be modulated through VO<sub>2</sub> film thickness.



*Figure 1.* (a) Temperature dependent Raman spectroscopic (473 nm laser) data taken from 25 to 100 °C for a 12.5 nm VO<sub>2</sub> films grown on native oxide Si at 200 °C. (b) SEM image of uncoated indium-tin oxide (ITO) nanorods and (c) SEM image of ITO nanorods coated with 12.5 nm of ALD VO<sub>2</sub> film. Scale bars are 1 $\mu$ m in (b) and (c).